


## Article

# Diffraction Features from $(10\bar{1}4)$ Calcite Twins Mimicking Crystallographic Ordering

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**Abstract:** During phase transitions the ordering of cations and/or anions along specific crystallographic directions can take place. As a result, extra reflections may occur in diffraction patterns, which can indicate cell doubling and the reduction of the crystallographic symmetry. However, similar features may also arise from twinning. Here the nanostructures of a glendonite, a calcite ( $\text{CaCO}_3$ ) pseudomorph after ikaite ( $\text{CaCO}_3 \cdot 6\text{H}_2\text{O}$ ), from Victoria Cave (Russia) were studied using transmission electron microscopy (TEM). This paper demonstrates the occurrence of extra reflections at positions halfway between the Bragg reflections of calcite in  $0kl$  electron diffraction patterns and the doubling of  $d_{104}$  spacings (corresponding to  $2 \cdot 3.03 \text{ \AA}$ ) in high-resolution TEM images. Interestingly, these diffraction features match with the so-called carbonate *c-type* reflections, which are associated with Mg and Ca ordering, a phenomenon that cannot occur in pure calcite. TEM and crystallographic analysis suggests that, in fact,  $(10\bar{1}4)$  calcite twins and the orientation change of  $\text{CO}_3$  groups across the twin interface are responsible for the extra reflections.



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**Keywords:** electron diffraction; *c-type* reflections; ordering; twinning; calcite; glendonite; TEM

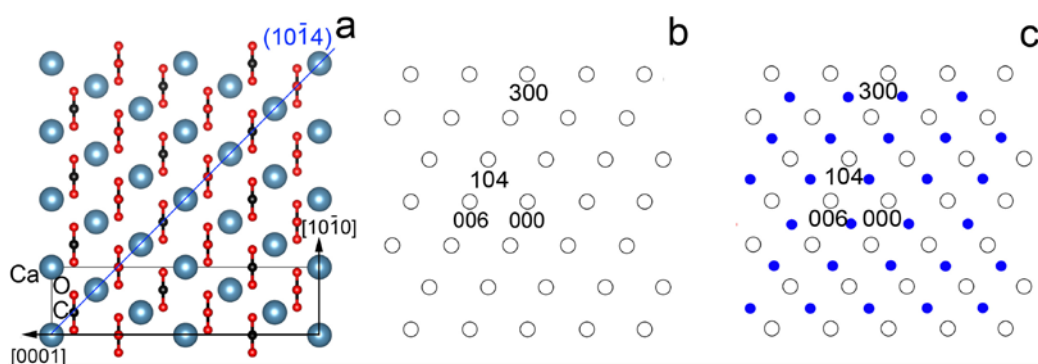
## 1. Introduction

Transmission electron microscopy (TEM) is an excellent method for characterizing the nanostructures of crystalline materials. A wide range of imaging and diffraction techniques provide unique information about the crystal structure of the studied samples, such as the local elemental composition, electronic structure and bonding down to the atomic level [1]. Thanks to the latest developments, not only has the resolution limit been pushed below  $0.5 \text{ \AA}$ , but nowadays even the behavior of materials and devices at the atomic scale can be mapped [2,3]. TEM is practically an indispensable technique for investigating crystal defects and studying phase transitions. However, the interpretation of TEM data can be challenging, and ambiguities can arise for cases that look simple [4]. In particular, the recognition of twins from TEM data can be difficult because their diffraction features can be confused with ordering and superstructures [5–7].

Here TEM is used to study the nanostructure of glendonite, which is a synonym for calcite pseudomorph after ikaite ( $\text{CaCO}_3 \cdot 6\text{H}_2\text{O}$ ), as a result of transformation. Glendonite is commonly associated with cold paleotemperature [8–10]. However, issues have been raised with regard to using it as an indicator for cold temperature [11] because, in a laboratory setting, ikaite formation was reported even above  $20 \text{ }^\circ\text{C}$  [11,12]. Ikaite rapidly disintegrates into a mush of water and recrystallizes to calcite during slight warming or pressure release, resulting in a highly porous crystal mesh [13–16]. Aragonite (unit cell parameters  $a = 4.9611 \text{ \AA}$ ,  $b = 7.9672 \text{ \AA}$ ,  $c = 5.7404 \text{ \AA}$ ; space group: *Pm $\bar{c}$ n*) has also been observed from ikaite transformation in marine and alkaline lake environments [17,18]. Studies on the transformation of synthetic ikaite suggest that it can also transform to amorphous calcium carbonate [19] and vaterite (unit cell parameters  $a = 4.13 \text{ \AA}$ ,  $b = 7.15 \text{ \AA}$ ,  $c = 8.48 \text{ \AA}$ ; space group: *Pbnm*) (e.g., [20]). Macroscopic calcite pseudomorphs after ikaite are

traditionally called glendonite, although other names also are used [8]. According to [11], the pseudomorph replacement of ikaite by calcite occurs through a coupled dissolution–reprecipitation mechanism at the ikaite–calcite interface. Since glendonites preserved the parent phase morphology, structural relicts indicating phase transition can be expected [21]. In fact, the TEM data here reported indicated an interesting lamellar structure with the occurrence of diffraction features that are inconsistent with ordinary calcite and that could mistakenly be associated with crystallographic ordering and superstructures.

Understanding the structural details of Ca-Mg carbonates is an important topic since they are rock-forming minerals. In particular, calcite is the dominant carbonate phase on the Earth’s surface and it is the third most abundant mineral after quartz and feldspars. It abundantly precipitates at ambient conditions and is one of the most important biominerals. Calcite is rhombohedral (unit cell parameters  $a = 4.989 \text{ \AA}$  and  $c = 17.06 \text{ \AA}$ ; space group:  $R\bar{3}c$ ). In this paper, three independent indices  $hkl$  are used to label reflections and  $d$ -spacings and four indices  $hkil$  ( $i = -(h + k)$ ) are used whenever crystallographic planes and directions are discussed. Ca atoms lie in the  $(10\bar{1}4)$  plane alternating with  $\text{CO}_3$  triangular groups along the  $[0001]$  (Figure 1a). The orientation change of the carbonate groups implies a  $c$  glide that is revealed by systematic absences ( $l = 2n + 1$  for  $0kl$  reflections) in the corresponding reciprocal lattice (Figure 1b). In dolomite,  $(10\bar{1}4)$  Mg and Ca cation layers alternate along  $[0001]$ ; as a result, the  $R\bar{3}c$  symmetry is reduced to  $R\bar{3}$  and  $b$ -type reflections ( $l = 2n + 1$  for  $0kl$  reflections) occur (e.g., [22]). Although such  $b$ -type reflections do not occur in pure conventional calcite, they are present in a disordered phase of calcite (space group:  $R\bar{3}m$ ) as a consequence of the rotational disorder of the  $\text{CO}_3$  groups. This polymorph was reported at high temperature ( $>1260 \text{ K}$ ) and at high  $p\text{CO}_2$  [23]. The reflections halfway between those of the  $a$ -type ( $l = 2n$  for the  $0kl$  reflections) and  $b$ -type are the so-called  $c$ -type reflections (Figure 1c), which have been associated with ordering and attributed to various superstructures in Mg-bearing calcite and dolomite [22,24,25]. In contrast to these explanations, Larsson and Christy [5] showed that the  $c$ -type reflections can arise from submicron-sized calcite twins. Shen et al. [26] followed this explanation and demonstrated that multiple diffraction between the host dolomite and twinned Mg-calcite nano-lamellae could give rise to the  $c$ -type reflections in Ca-rich dolomite.



**Figure 1.** (a) Structure of calcite seen along  $[01\bar{1}0]$ . In order, the large blue circles and red and black dots represent Ca, O and C atoms. The  $(10\bar{1}4)$  plane is marked with a blue line. (b)  $0kl$  diffraction plane of  $R\bar{3}c$  calcite with  $a$ -type reflections (empty circles). (c)  $0kl$  diffraction plane with  $c$ -type reflections (blue dots) occurring halfway between  $a$ -type reflections.

The aim of this paper is to confirm Larsson and Christy’s [5] hypothesis via the recognition of calcite  $(10\bar{1}4)$  twins from selected-area electron diffraction (SAED) patterns and high-resolution TEM images (HRTEM). Here it is demonstrated that the  $c$ -type reflections in glendonite can be successfully interpreted in terms of  $(10\bar{1}4)$  twinning by reticular pseudo-merohedry [27]. It is also shown that streaked  $c$ -type reflections can be associated with  $(10\bar{1}4)$  stacking faults occurring inside the small (5–15 nm wide) twin domains. The proper interpretation of these diffraction features is crucial as they can lead to erroneous conclusions.

## 2. Materials and Methods

Glendonite samples from Victoria Cave (Bashkiria, Russia) were received from Paul Töchterle and Yuri Dublyansky (Institute of Geology, University of Innsbruck). The description of the site and the characteristics of the samples are reported in [28] and [21] respectively.

Glendonite grains were coated in gold and measured using a Zeiss EVO 40 scanning electron microscope operated at 20 kV. A back-scattered electron (BSE) image was acquired at 10,000 magnifications and the chemical composition of the sample was measured using an Oxford INCA energy dispersive X-ray (EDX) spectrometer.

Glendonite powder was analyzed with an X-ray diffraction (X'Pert, PANalytical B.V.) method with Cu K $\alpha$  radiation in the  $2\theta$  range of 10 to 60° and with a 0.04° step scan for 1 sec. Si was used as an internal standard. The unit-cell parameter of calcite was refined based on the 012, 104, 110, 113, 202, 024 and 116 reflections using the PDIndexer program [29].

Zoltán May (ELKH Research Centre for Natural Sciences, Budapest) determined the elemental composition of glendonite by atomic emission spectroscopy using a Spectro Genesis ICP-OES (SPECTRO Analytical Instruments GmbH, Kleve, Germany) simultaneous spectrometer with axial plasma observation. Multi-element standard solutions for ICP (Merck Chemicals GmbH, Darmstadt, Germany) were used for calibration. The detection limits of the analytical method are reported in [30].

For TEM investigation the samples were crushed in ethanol and deposited onto copper grids covered by Lacey carbon supporting films. Bright-field TEM (BFTEM), HRTEM and SAED data were acquired with a 200 keV Talos Thermo Scientific electron microscope. Fast Fourier transforms (FFTs) obtained from the HRTEM images were calculated using Gatan Digital Micrograph 3.6.1 software. Structure models of calcite were drawn using VESTA-win64 software [31] and the twinning features of calcite were tested with the software GEMINOGRAPHY [32].

## 3. Results and Discussion

### 3.1. *c*-Type Reflections and the Twin Lattice

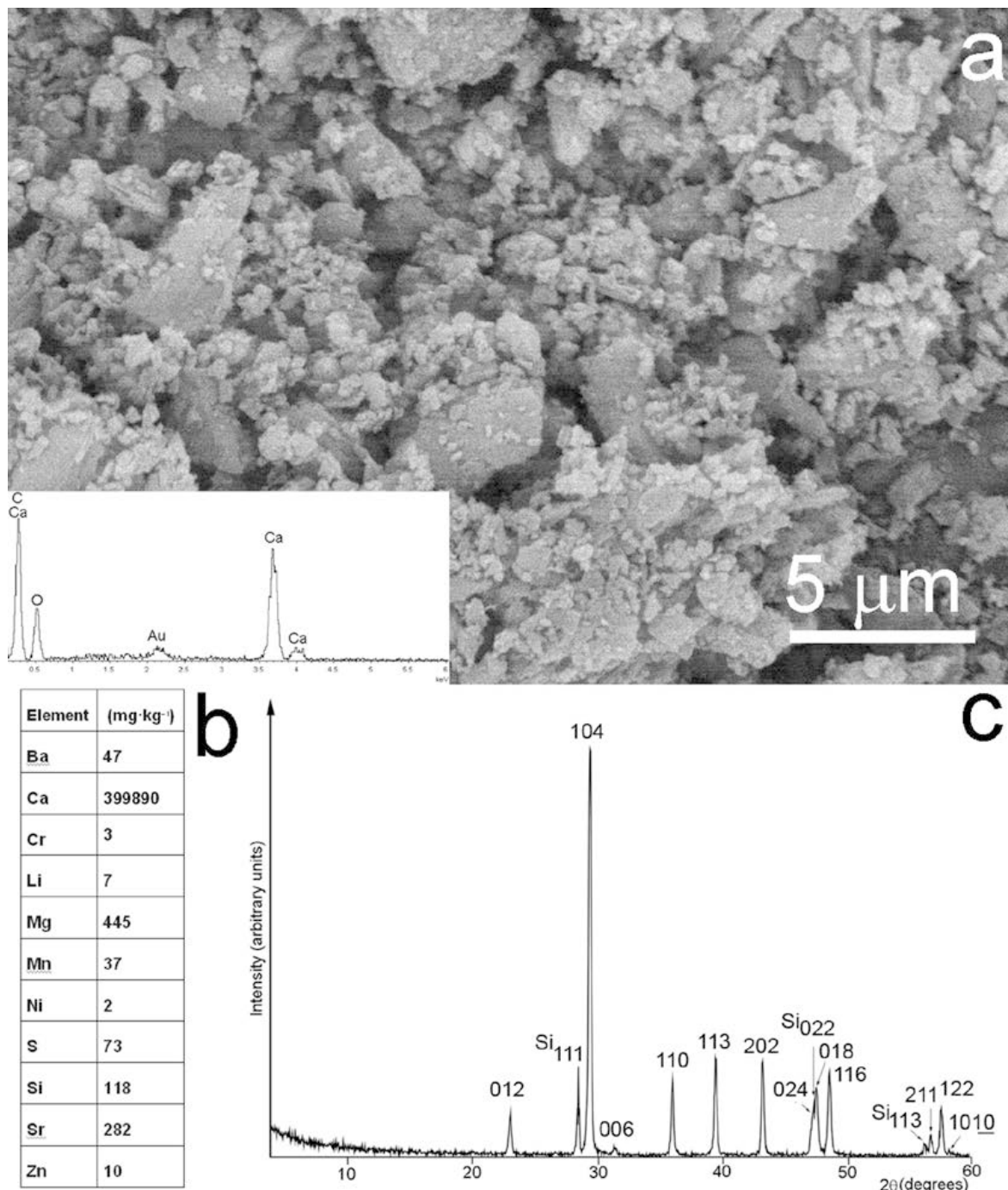
The studied glendonite is built up of fine-grained (submicrons to several microns in size) calcite crystals (Figure 2a). The sample is chemically homogeneous, practically pure CaCO<sub>3</sub> (Figure 2a,b). Its water content is negligible [21] and its X-ray diffraction pattern is consistent with ordinary calcite, having the unit cell parameters  $a = 4.989$  (2) and  $c = 17.08$  (1) (Figure 2c). However, TEM reveals that the sample has a complex nanostructure. In particular, BFTEM images taken along [01 $\bar{1}$ 0] show linear features parallel to (10 $\bar{1}$ 4) and indicate the occurrence of 10–15 nm size nano-domains in the calcite matrix (Figure 3a). SAED patterns of the nano-domains reveal weak *c*-type reflections that are incompatible with the  $R\bar{3}c$  space group of calcite (Figure 3b).

It is intriguing that the *c*-type reflections of the studied sample mimic the diffraction features of ikaite ( $a = 8.792$  Å,  $b = 8.310$  Å,  $c = 11.021$  Å,  $\beta = 110.53^\circ$ , space group:  $C2/c$ ) and also those of a hypothetical carbonate supercell containing ordered cation sites. In fact, the measured  $d$  spacings of 6.05 Å and 3.85 Å roughly match the  $d_{110}$  (5.849 Å) and  $d_{2-1-1}$  (3.882 Å) of ikaite. In addition, the calculated angle between (110) and (2 $\bar{1}$ 1) is 75.7°, which is close to the measured value of 73°. However, ikaite loses its water content in vacuum within seconds; thus, it cannot be studied with TEM. Furthermore, Fourier transform infrared spectroscopy does not indicate a detectable amount (<1 wt%) of crystalline water inside the sample [21]. Similarly, the interpretation of *c*-type reflections with cation ordering is not viable, because both EDX and ICP-OES analysis (Figure 2) suggest that the sample does not contain a detectable amount (<1 wt%) of foreign cations.

In contrast to the abovementioned hypothetical explanations, (10 $\bar{1}$ 4) calcite twins provide a convincing solution for the diffraction features of the SAED pattern (Figure 3b). In particular, twin individuals hosted in an underlying calcite lattice [5] give rise to the observed extra and split reflections (Figure 3c). According to tests using the software GEMINOGRAPHY [32], the observed twin corresponds to a (10 $\bar{1}$ 4) twin by reticu-

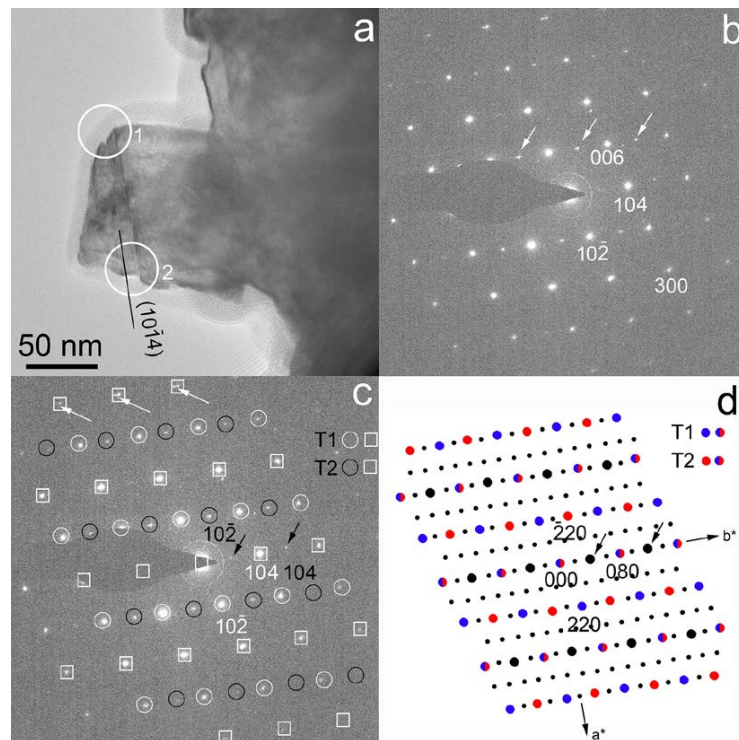
lar pseudo-merohedry (twin index 4 and obliquity  $0.74^\circ$ ) based on a pseudo-orthorhombic C-centered sublattice with cell parameters  $a = 8.095 \text{ \AA}$ ,  $b = 24.286 \text{ \AA}$ ,  $c = 4.989 \text{ \AA}$ ,  $\alpha = 90.00^\circ$ ,  $\beta = 90.00^\circ$  and  $\gamma = 89.26^\circ$ . The calculated obliquity ( $0.74^\circ$ ) is in accordance with the small observed split of reflections in the SAED pattern (Figure 3c). The relationship between the unit cell of calcite and that of the pseudo-orthorhombic twin lattice is described by the following matrix T:

$$\begin{pmatrix} 4/3 & 2/3 & -1/3 \\ 4 & 2 & 1 \\ 0 & -1 & 0 \end{pmatrix}$$

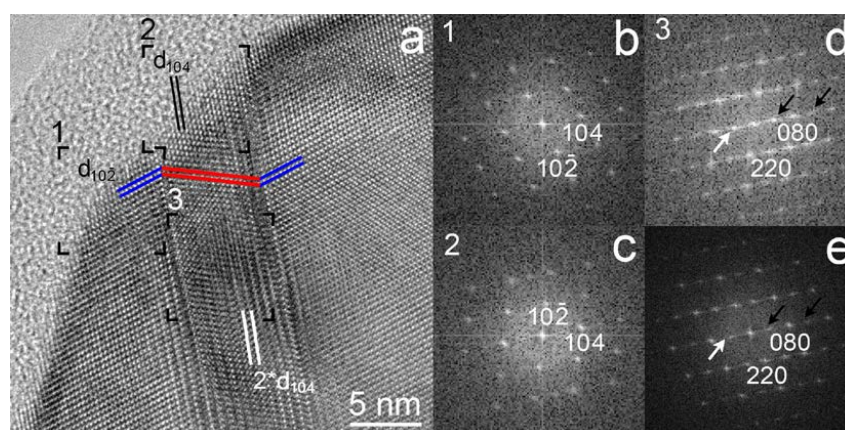


**Figure 2.** (a) BSE image and EDX data of the studied glendonite. (b) Elements present in glendonite. (c) X-ray powder diffraction pattern of glendonite calcite. Si standard was added for measuring the cell parameters.

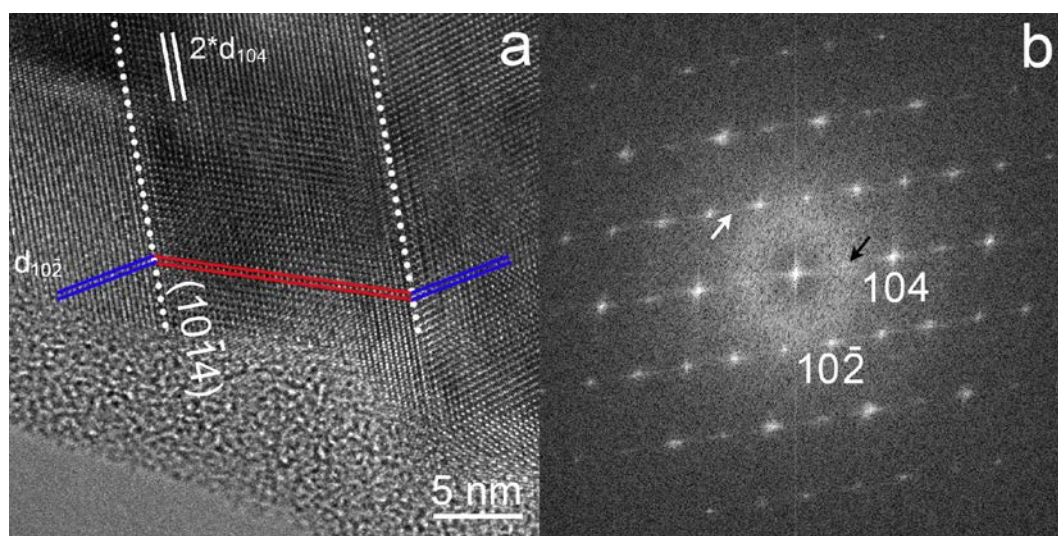




**Figure 3.** (a) BFTEM image of a  $\sim 10$  nm size  $(10\bar{1}4)$  twinned domain within the calcite matrix. The areas 1 and 2, marked by white circles, are magnified in Figures 4 and 5, respectively. (b) The SAED pattern along  $[01\bar{1}0]$  obtained from a  $\sim 200$  nm size region shown in (a) reveals weak *c*-type reflections (white arrows). (c) The interpretation of the SAED pattern (b) via  $(10\bar{1}4)$  twinning. The splitting of the overlapping reflections (white squares) increases with increasing diffraction angle (white arrows). Black arrows mark reflections arising from dynamically scattered electrons of the  $(10\bar{1}4)$  twin hosted within a calcite matrix. (d) The interpretation of the SAED pattern (b) with the pseudo-orthorhombic twin cell ( $hk0$  diffraction plane). Large-size black dots denote the reflections marked by black arrows in (c).



**Figure 4.** (a) HRTEM image of a  $\sim 10$  nm size  $(10\bar{1}4)$  twinned domain within calcite along  $[01\bar{1}0]$  (area shown in Figure 3a with white circle 1). Blue and red lines show  $d_{10-2}$  spacings of three polysynthetic twinned individuals. Black and white lines show single and doubled  $d_{104}$  spacings, respectively. FFTs calculated from regions 1 (b) and 2 (c) correspond to twin individuals 1 and 2. (d) FFT calculated from region 3 corresponds to an area where individual 2 is hosted in the calcite matrix. (e) FFT calculated from the entire HRTEM image (a) shows a diffraction pattern similar to inset (d). Dynamically scattered electrons arising from the thick part of the sample result in reflections marked by black arrows in (d) and (e). The FFTs of (d) and (e) are interpreted with the pseudo-orthorhombic twin cell. White arrows mark the streaked reflections for (d) and (e), which are indicative for  $(10\bar{1}4)$  stacking faults.



**Figure 5.** (a) HRTEM image of a ~15 nm size  $(10\bar{1}4)$  twin domain within calcite along  $[01\bar{1}0]$  (area shown in Figure 3a with white circle 2). Blue and red lines mark  $d_{10-2}$  spacings of the twin individuals. White lines mark doubled  $d_{104}$  spacings. (b) FFT calculated from the HRTEM image (a) showing *c-type* reflections (black arrow). The FFT is indexed according to rhombohedral calcite. The white arrow shows streaked reflections, which can indicate  $(10\bar{1}4)$  stacking faults.

Applying this matrix to any  $[u, v, w]$  vectors of calcite, the corresponding  $[u', v', w']$  vectors of the pseudo-orthorhombic cell can be generated. In addition, applying this matrix to indexes of the calcite  $hkl$  reflections, we can transform them to the  $h'k'l'$  indexes referred to the twin lattice. For example, the 104 reflection of calcite corresponds to 080 of the pseudo-orthorhombic twin lattice. The twin lattice explains all the observed reflections and shows that every fourth set of reflections of the individuals overlaps on the  $hk0$  plane (Figure 3d), as expected from the twin index 4.

### 3.2. HRTEM Images of the $(10\bar{1}4)$ Calcite Twins and Doubled $d_{104}$ Spacings

The identification of the  $(10\bar{1}4)$  calcite twins is straightforward from the HRTEM image acquired with the electron beam incident along  $[01\bar{1}0]$  of the thin part (~10 nm thick) of the sample; in fact, the two individuals can be easily recognized by the orientation change of the fringes corresponding to  $d_{012}$  spacings (3.85 Å) and they can be localized from their corresponding FFTs (Figure 4a–c). The HRTEM image (Figure 4a) obtained from the thin part also shows fringes with 3.03 Å corresponding to  $d_{104}$  spacing. However, 6.05 Å spacing can be measured from the thicker areas (>10 nm) of the image (white lines in Figure 4a) and its corresponding FFT (Figure 4d), as well as the FFT calculated from the entire HRTEM image (Figure 4e). This  $d$ -spacing, which is also evident in Figure 5a, erroneously indicates cell doubling.

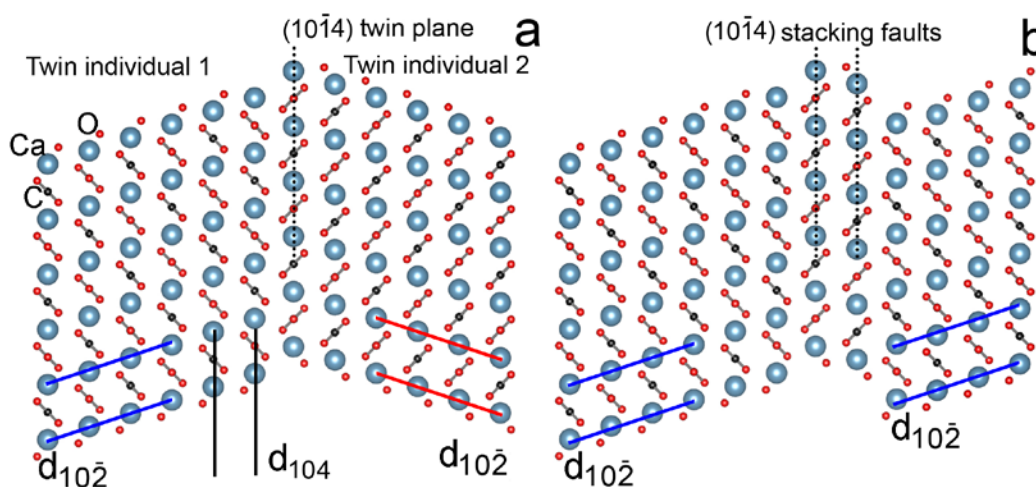
It should be noted that the systematic absences of the  $R\bar{3}c$  space group, i.e., those of the twin individuals, should be present even for thick samples. Thus, the side-by-side arrangement of the individuals does not provide an explanation for the *c-type* reflections of the FFT calculated from the thick part of the grain (Figure 4d). Therefore, following the suggestion of Larsson and Christy [5], it is proposed that the small (<10 nm) twin individual 2 is actually hosted in the calcite matrix (individual 1). Since HRTEM images are two-dimensional projections of the three-dimensional structures, the HRTEM image acquired with the electron beam incident along  $[01\bar{1}0]$  shows the vertical projection of the individual 2 and that of the underlying calcite (individual 1) from areas thicker than the individual twin size (>10 nm). It is suggested that electrons dynamically scattered from these vertically stacked twin individuals give rise to the doubled  $d_{104}$  spacings (2·3.03 Å) of the HRTEM image and the *c-type* reflections occurring along  $[104]^*$  of the FFTs (Figure 4d,e and Figure 5b). In fact, Christy and Larson [5] have demonstrated that, by

considering twins in an underlying calcite matrix, even the doubled  $d_{012}$  corresponding to a 2.3.85 Å spacing can be generated. Thus, the doubled  $d_{104}$  and the  $d_{012}$  spacings measured on the thick part (>10 nm) of the HRTEM images can be used as evidence for nanosized calcite twins.

Nanotwins hosted in a matrix are commonly observed, in particular for materials such as diamond and SiC that contain abundant stacking faults [6,7,33,34]. In fact, the FFTs calculated from the thick region and the entire image show streaked reflections (white arrows in Figure 4d,e and Figure 5b), which can be associated with  $(10\bar{1}4)$  stacking faults that occur inside the 5–15 nm wide twin individuals.

### 3.3. Crystal Structure of the $(10\bar{1}4)$ Calcite Twins

The crystallographic analysis suggests that the *c*-type reflections in calcite and their streaking can be successfully interpreted via  $(10\bar{1}4)$  twins and stacking faults. The structure model of this twin along  $[01\bar{1}0]$  (Figure 6a), built based on the work of Bruno et al. [35,36], reveals the orientation change of the  $\text{CO}_3$  groups and the  $(10\bar{1}2)$  calcite planes across the twin interface, which agrees well with the observed features (Figures 4a and 5a). Based on geometry optimizations of 2D twinned slabs, Bruno et al. [35,36] demonstrated that the twinned individuals are translated and, at the twin interface, tilting and rotation of the  $\text{CO}_3$  groups occur, which gives rise to variations of the Ca–O distances. Following the construction of the twin model, the crystal structure of a  $(10\bar{1}4)$  stacking fault can also be generated (Figure 6b).



**Figure 6.** (a) Structure model of  $(10\bar{1}4)$  twin (a) and  $(10\bar{1}4)$  stacking faults (b) constructed based on the work of Bruno et al. [35,36]. In order, blue, red and black dots represent Ca, O and C atoms.

### 3.4. Possible Origin of the $(10\bar{1}4)$ Calcite Twins

The  $(10\bar{1}4)$  twin is one of the four possible twins observed in calcite [37]. Although it has been reported as one of the main deformation twinning planes of calcite [37], it is also commonly associated with crystal growth and has been reported among others from the calcite shell of a sea urchin [5]. This twin was observed in a glendonite sample that formed from ikaite; thus, it may be associated with the ikaite-to-calcite transformation. However, since the formation of aragonite [17,18], amorphous calcium carbonate [19] and vaterite [20] has also been reported during this transition, a multiphase origin of the calcite twin is also plausible. In either way, interpretation based on the occurrence of this twin alone is definitely not a diagnostic for identifying glendonites and inferring cryogenic conditions [21].



#### 4. Conclusions

*c-type* reflections (Figure 1) were identified in glendonite, a calcite pseudomorph after ikaite. As the sample was chemically pure  $\text{CaCO}_3$  (Figure 2), it turned out that the *c-type* reflections cannot mimic the diffraction features associated with crystallographic ordering. TEM and crystallographic analysis suggested that, in fact, twinning by reticular pseudomohedry (twin plane:  $(10\bar{1}4)$ ; twin index: 4; obliquity:  $0.74^\circ$ ) successfully explains these reflections (Figure 3). The twin individuals could be recognized from thin areas ( $\sim 10$  nm) of the HRTEM images through the direction change of the fringes corresponding to  $d_{10-2}$  spacings ( $3.85\text{Å}$ ). From the thicker part of the sample ( $>10$  nm), the small-sized twin individual (5–10 nm) hosted in the calcite matrix and dynamically scattered electrons gave rise to double  $d_{104}$  spacings corresponding to  $2.3.03\text{Å}$  measured on HRTEM images (Figures 4 and 5). Streaked *c-type* reflections were also observed, and they were explained by  $(10\bar{1}4)$  stacking faults occurring inside the small (5–15 nm wide) twin domains. It was shown that the orientation change of carbonate groups across the  $(10\bar{1}4)$  calcite planes (Figure 6) gave rise to the  $(10\bar{1}4)$  twins and stacking faults. Although in the glendonite sample Mg and Ca ordering could not occur, this phenomenon may be present in Mg-containing calcite, from which *c-type* reflections have been reported. This paper strongly supports (1) the interpretation of Larsson and Christy [5] associating *c-type* reflections with  $(10\bar{1}4)$  twins and with the orientation change of carbonate groups across the twin interface and (2) the final conclusion of considering the occurrence of  $(10\bar{1}4)$  twins before attributing *c-type* reflections to Mg and Ca ordering.

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**Conflicts of Interest:** The author declares no conflict of interest.

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