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# The Abrasion Behaviour of X40CrMoV5-1 Steel Under Various Surface Treatments

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#### Abstract

For evolving a surface layer on the X40CrMoV5-1 steel, a plasma-nitridation and PVD coating process was applied. In our experiments, the samples were heat-treated (high-temperature hardening, annealed three times) and surface treatments (plasma-nitridation, PVD coating by TiAlN, duplex surface treating by plasma nitridation and after that, PVD coating TiAlN). After the heat treatments, we performed hardness tests and surface abrasion wear tests. The abrasion wear resistance of the specimens was studied in order to understand the heat treatment effects on abrasion behaviour. It was observed that the heat treatment and surface treatment process greatly influence the tool steel surface hardness and abrasion resistance behaviour. By plasma-nitridation the surface hardness doubled compared to the quenched surface hardness while the

PVD coated TiAlN surface layer hardness is more than five times that of the hardened one. There was no relevant difference between the PVD coated (TiAlN) surface hardness and the duplex surface-treated hardness. On the basis of the results of the comparative abrasive wear tests, it can be concluded that the duplex surface treatment resulted in the greatest wear resistance.

Keywords: tool steel, secondary hardening, austenitization, PVD coating, hardness, plasma nitriding.

## 1. Introduction

It is characteristic of hot-forming tool steels that they often have to meet opposing requirements at the same time, i.e. they have to function under the combined effect of different stresses. It follows that the main properties of the hot forming tool steels are hot strength, hardness, abrasion resistance, toughness, annealing resistance, thermal fatigue resistance, hot workability and machinability. These properties can be achieved partly by alloying with different chemical elements and partly by applied special heat treatment technologies. Depending on their use, there are cases where the toughness and abrasion resistance of the material are required at the same time. In such cases, surface treatments are used. The abrasion resistance, high strength and toughness can all be achieved at the same time by applying a thin wear-resistant layer on the surface of the material. This is possible in several ways.

The X40CrMoV5-1 steel belongs to the family of hot working tool steels which is known among tool makers as EN 1.2344 The material has a very good heat resistance and hot wear resistance [1, 2]. These steels retain their properties even when working at temperatures of up to 200 °C, due to the optimal heat treatment technology and the chemical composition of the material. An increase in surface abrasion resistance is achieved by creating a thin abrasion-resistant layer [3, 4]. This can be achieved by nitriding or/and coating on hot forming tool steels. Plasma nitriding technology was introduced in 1920, but in industrial applications has been only used for the past 30 years. The plasma nitriding technology is a thermochemical surface treatment process which is carried out at a temperature of 350–600 °C. In front of the furnace wall serving as an anode, positive ions collide at high impact velocities with workpieces connected to the cathode. This ionic shower first produces a very intensive surface cleaning and then heats and nitrides the workpiece surface [5].

Plasma nitridation can be performed in both DC and pulsed plasma. Plasma nitridation takes place in a vacuum furnace (200 to 500 Pa pressure), in an ionized gas atmosphere (ammonia, nitrogen, methane or hydrogen), but gas mixtures are also used to form more abrasion resistant surface layers.

The quality of the heat treatment is determined by the composition of the gas, the pressure, the temperature and the duration of the operation. The upper part of the nitride layer (up to 30  $\mu$ m depending on the material) is hard and chemically stable, and below it is the diffusion zone with max. 1 mm thickness.

In the case of hot-forming tool steel, a surface hardness of 850–950 HV can be achieved up to a depth of 0.4 mm by plasma nitriding. The process improves the abrasion and sliding properties of the materials and creates a corrosion-resistant layer on the material surface. The dimension distortion is very small. Plasma nitriding is usually used as the final process in finished parts because after this treatment no machining is required.

The dimensions of nitrided workpieces increase by ~1 % of the bark thickness, which is an insignificant change. This procedure allows precise control of the process, fine-tuning of the microstructure of the nitride layer, and thus achieving the desired properties. PVD surface treatment is a physical vapour deposition in which the coating material is applied to the surface of the coating component in atomic, chemical compounds or ionic form. PVD processes take place at relatively low temperatures, which means that the process does not modify the tissue structure of the sublayer and there is no allotropic transformation so that the surface-treated tool does not suffer any dimensional changes.

The PVD coating provides good sliding properties, high abrasion resistance and high hardness on the tool surface. The main purpose of the surface treatments was to increase the wear resistance of the tool. The abrasive wear is a complex system of microscopic interactions that occur between sliding surfaces. These interactions depend on the contact surfaces of the materials, on their physical, chemical, mechanical properties, of the geometry, and external conditions that affect the wear (e.g., temperature) [6–7].

Wear processes can only be investigated in a well-defined wear system, for comparison, measured results that have been determined in a similar tribological system and where only the investigated parameter changes, other parameters are minimized [8–9]. Tool wear resistance is not easy to predict. Hardness measurement is used in many places in the industry for this purpose, as there is some correlation between abrasion resistance and hardness. Similarly, comparative experimental methods are widespread, e.g. ball / flat abrasion test [10–12]. During our measurements, we examined the wear resistance of the specimens with different heat treatment and surface treatment by a comparative method and also compared their hardness values.

# 2. The experimental materials and technologies

The chemical composition of the tested X40CrMoV5-1 steel specimens is shown in Table 1.

Table 1. Chemical composition (%)

С	Si	Mn	Cr	Мо	v
0.40	1.10	0.45	5.25	1.41	1.2

The austenitization of the samples was performed in a VFC type 300×370×200 mm Ipsen vacuum furnace with two-stage heating (650 °C and 850 °C, respectively) at 1050 °C. The cooling was performed with 6 bar nitrogen gas. The quenching was followed by triple high-temperature tempering (**Figure 1**.).



Figure 1. Az X40CrMoV5-1 steel heat treatments diagram.

Tempering was performed in a 300×370×350 mm Muehl furnace under argon gas each with two hours heating times.

After each guenching and tempering operation, Vickers hardness measurements were performed according to standard practice [13–14]. This was done on a Buhler 1105 machine. After that, samples 2 and 4 were plasma nitrided. The cleaning was performed in a mixture of hydrogen (40 L/h), argon (5 L/h) and nitrogen (1 L/h) inert gas. After that, the specimens were plasma nitrided at 480 ° C for 24 hours in a mixture of hydrogen (120 L/h) and nitrogen (40 L/h) gas. After cooling, Vickers hardness measurements were performed. Samples 3 and 4 were subjected to a TiAlN PVD coating process, which was also followed by a Vickers hardness test. The final test for all four specimens was the abrasive wear test. Ball wear equipment was used for wear testing (Figure 2.). Before the wear test, the roughness was measured using the roughness measuring equipment shown in Fig**ure 3.** The abrasion ball was a 20 mm diameter  $Al_2O_3$ -based ceramic ball with a polished surface. The abrasion factor (K) (1), which is calculated from the abrasion volume (*V*,), the abrasion path length (S) and the load force (N), was used as the wear indicator.

$$K = \frac{v_v}{s_N}, \quad \left(\frac{mm^3}{N \cdot m}\right) \tag{1}$$

The abrasion volume is calculated from the diameter of the abrasion trace (*d*) and the depth of the spherical bottle (*h*) (2):

$$V_{v} = \frac{h\pi}{6} \left(\frac{3}{4}d^{2} + h^{2}\right), (mm^{3})$$
<sup>(2)</sup>

The depth of the abrasive sphere is calculated by a simple relationship between the radius R of the abrasion ball and the diameter of the abrasion imprint (d = 2R) (3).

$$h = R - \sqrt{R^2 - \left(\frac{d}{2}\right)^2} , \quad (mm)$$
<sup>(3)</sup>

The wear path length (S) depends on the wear time (t), on the radius of the wear sphere and its speed (n) (4):

$$S = n \cdot 2 \cdot \pi \cdot R \cdot t \quad , \qquad (m) \tag{4}$$

To be comparable with the data in the literature, the speed was set at 570 rpm and the abrasion test time was 5 minutes.



Figure 2. Ball wear equipment.



Figure 3. Mahr roughness tester.

#### 3. Test results

In the delivery condition, the average hardness of the tool steel to be tested (X40CrMoV5-1) was 215 HV.

#### 3.1. Effects of the heat treatments (Table 2.)

After quenching from 1050 °C and three times tempering at 520 °C the hardness of the specimen was 549HV. After plasma nitriding, the hardness of the specimens 2 and 4 was 1140 HV. Sample 3, after the vacuum hardening was PVD coated with TiAlN coating and reached a surface hardness of 2938HV. Sample 4, after plasma nitriding, was PVD coated with TiAlN coating, so it was duplex surface treated and the result was a surface hardness of 2539 HV.

Table 2. The hardness after heat treatments

Sample	Heat treatment	Coating thickness (µm)	Hardness (HV)
1	Quenched and tempered	0	549
2	Quenched, tem- pered and plasma nitrided	0	1140
3	Quenched, tem- pered and TiAlN coated	2.05	2938
4	Quenched, tem- pered, plasma nitrided and TiAlN coated	1.93	2539

### 3.2. Effects of the heat treatments

The abrasion factor calculated from roughness measurements and formula (1) is shown in the **Table 3.** 

After the wear abrasion test, the diameters of the worn craters were measured with an optical microscope type Neophot 2. (Figure 4.).

### 4. Conclusions

The abrasion resistance results of the examined samples from material quality X40CrMoV5-1, treated with various heat treatments and surface treatments, are shown in Figure 5.

Based on the results, conclusions are as follows:

- I. The TiAlN-based PVD coating showed the highest abrasion resistance based on experimental methods and parameters.
- II. The hardness and wear factor of the test specimen with a plasma nitrided or PVD coated surface also resulted in much more favourable properties than the surface with only a hardened and tempered conventional heat treatment.
- III. Plasma nitrided and PVD coated surfaces, however, have a lower hardness than hardened-welded and PVD coated surfaces resulting in better abrasion resistance. However, for this result to be accurately evaluated, it is important that the surface roughness of both specimens be identical.

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Table 3. Roughness and wear factor values

Sam- ple	Heat treatment	Ra (µm)	Wear factor (mm³/(N·m))
1	Quenched and tempered	0.010	6.32×10 <sup>-9</sup>
2	Quenched, tem- pered and plas- ma nitrided	0.057	1.95×10 <sup>-9</sup>
3	Quenched, tem- pered and TiAlN coated	0.233	8.46×10 <sup>-10</sup>
4	Quenched, tem- pered. plasma nitrided and TiAlN coated	0.177	7.57×10 <sup>-10</sup>



Figure 4. Wear imprint of sample nr. 3.



Figure 5. Measurement results: Effect of heat treatment and surface treatment technologies on the abrasion resistance of steel.

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