Magnetic hyaluronate hydrogels: preparation and characterization

Ildikó Y. Tóth a, Gábor Veress a, Márta Szekeres a, Erzsébet Illés a, Etelka Tombácz a*

a Department of Physical Chemistry and Materials Science, University of Szeged, Aradi Vt. 1, H-6720 Szeged, Hungary
* Corresponding authors: Ildikó Y. Tóth (+3662544212, Ildiko.Toth@chem.u-szeged.hu) and E. Tombácz (+3662544212, tombacz@chem.u-szeged.hu)

Highlights
* Novel hyaluronate(HyA)-based biocompatible magnetic hydrogels were prepared.
* Chondroitin-sulfate-A coating is needed to disperse magnetite particles in HyA-gel.
* Rheological behaviour of hydrogels was independent of the magnetite content (<2 g/L).
* Gels remained in stable and homogeneously dispersed state even after 90 days storage.
* Magnetic HyA-gels are promising candidates for use as intra-articular injection.

Abstract
A novel soft way of hyaluronate (HyA) based magnetic hydrogel preparation was revealed. Magnetite nanoparticles (MNPs) were prepared by co-precipitation. Since the naked MNPs cannot be dispersed homogenously in HyA-gel, their surface was modified with natural and biocompatible chondroitin-sulfate-A (CSA) to obtain CSA-coated MNPs (CSA@MNPs). The aggregation state of MNPs and that loaded with increasing amount of CSA up to 1 mmol/g was measured by dynamic light scattering at pH~6. Only CSA@MNP with ≥0.2 mmol/g CSA content was suitable for magnetic HyA-gel preparation. Rheological studies showed that the presence of CSA@MNP with up to 2 g/L did not affect the hydrogel's rheological behaviour significantly. The results suggest that the HyA-based magnetic hydrogels may be promising formulations for future biomedical applications, e.g. as intra-articular injections in the treatment of osteoarthritis.

Keywords: magnetite, chondroitin-sulfate-A, hyaluronate, magnetic hydrogels, surface modification, intra-articular injection

1. Introduction
Hydrogels are in the focus of scientific interest because of their tuneable mechanical, chemical and biocompatible properties [1,2]. These materials have been widely used in tissue engineering and drug delivery [1,3-5]. Magnetic hydrogels have also been produced by filling hydrogels with magnetic particles (e.g. maghemite, magnetite, cobalt-ferrite) [1], this procedure makes them more suitable for certain biomedical applications, such as drug delivery [1,6,7], enzyme immobilization [1,8-10] or cancer therapy by the combination of controlled drug release and magnetic hyperthermia [1,11]. Magnetic hydrogels have been prepared by three, fundamentally different methods: blending, in situ precipitation and grafting-onto methods [1]. Innumerable magnetic hydrogels have been synthesized, e.g. superparamagnetic magnetite nanoparticles (MNPs) in kappa-carrageenan nanospheres [12], in situ synthesized MNPs in chitosan hydrogel [13] and magnetic hydrogel of chitosan and MNPs for a potential cancer therapy by the combination of controlled drug delivery prepared from (3-aminopropyl)trimethoxysilane-functionalized magnetite or cobalt-ferrite bound chemically to HyA by N-hydroxysuccinimide and N-(3-dimethylaminopropyl)-N-ethylcarbodiimidehydrochloride [15]. Injectable in situ forming hydrogel has been synthesized...
from oleic acid-coated iron oxide nanoparticles dispersed in toluene and hyaluronic acid modified with hydrazide and thiol groups (HA-hy-SH) for MRI contrast agent [16]. The giant HyA molecules flocculate MNPs and loose aggregates are formed, which can be prevented only by the pre-modification of either the MNP’s surface [15,17] or the chemical structure of HyA [18] or by the combination of the previous two [16,19]. Usually harmful chemicals were used during these preparations. In our laboratory, several types of core-shell magnetite nanoparticles have been produced [20-22] in the absence of risky chemicals by using poly(acrylic acid) (PAA) [23], poly(acrylic acid-co-maleic acid) (PAM) [24] and chondroitin-sulfate-A (CSA) [25] as coating agents to modify the MNP surface.

The osteoarthritis (also called degenerative joint disease) is a common form of arthritis, which causes health problems for millions of people [26]. There are many reasons potentially responsible for the osteoarthritis, such as genetic factors, increased stress (i.e., hard physical work, sport, overweight) and decreased viscoelasticity of the intra-articular fluid. The latter can be caused by the reduced amount of hyaluronate (HyA) in the synovial fluid [27]. The supplement of synovial fluid by hyaluronate-containing intra-articular injection is a common treatment for osteoarthritis [26-30]. The typical composition of these gels (for example Hyalgan, Ostenil-Tendon, SportVis, Synvisc, Synocrom) is an isotonic solution (150 mM NaCl, pH ~7.3 adjusted by buffer) containing 1% of sodium hyaluronate. There are several problems with this treatment, for example the rapid enzymatic degradation of HyA within the joints [31], the pain and the risk of infection during the injection procedure. Furthermore, during movement and under the weight of the patient’s body, the synovial fluid could be squeezed outside the joints reducing the efficiency of the intra-articular treatments. Our approach is to overcome the latter defect by using magnetic hydrogel and its flowability that can be regulated by an implanted permanent magnet.

The aim of our research was to prepare HyA-based magnetic hydrogel samples in a soft way using biocompatible chemicals, which may be potentially used as an intra-articular injection in the future. The chemical structures of hyaluronate and biocompatible chondroitin-sulfate-A are very similar (Fig. 1), so it is presumable that the surface of CSA-coated superparamagnetic magnetite nanoparticles (CSA@MNPs) is compatible with the HyA-gel. We intend to prepare stable MNPs and CSA-coated MNPs, and to test their stability and aggregation in dilute HyA-gels. Finally, we plan to prepare HyA-based magnetic hydrogels and to test the effect of increasing CSA@MNP concentration on the rheological properties.

2. Materials and methods
2.1. Materials

The magnetite was prepared by co-precipitation [32-36]. The FeCl₂, FeCl₃ and NaOH for magnetite synthesis were analytical grade reagents obtained from Molar, Hungary. The precipitate was purified carefully through washing and dialysis. It was stored at pH~3 and 4 °C as stable sol. The resulting iron-oxide was identified as magnetite, based on the characteristic black colour, strong magnetism and the X-ray diffraction pattern (JCPDS database [37]). The mean diameter of the MNPs, determined by Scherrer equation and transmission electron microscopy, was ~10 nm [27].

The chondroitin-sulfate-A (CSA) was purchased from Sigma-Aldrich as sodium-salt, and the sodium-salt of hyaluronic acid (HyA) was kindly provided by the Department of Pharmaceutical Technology (University of Szeged, Hungary). One repeating unit of CSA (M=503 Da) contains one –COOH and one –SO₃H group (Fig. 1a) and that of HyA (M=379 Da) contains one –COOH group (Fig. 1b). The strongly acidic sulfate groups (–SO₃⁻) in CSA are fully deprotonated at a wide pH-range [38,39]. However, the –COOH groups in CSA and HyA have pH-dependent dissociation (pKᵢ of glucuronic acid ~2.9) [40] but, even so, these groups are fully deprotonated (–COO⁻) at pH ~6.3 applied in our experiments. The notations “CSA” and
"HyA" are used in this article for sodium-salt regardless of the actual degree of dissociation of the carboxylic groups. The amount of CSA and HyA is expressed through the mole of repeating units, which equals to the number of dissociable –COOH groups.

The CSA-coated magnetite nanoparticles (CSA@MNP)s were prepared at pH=6.3±0.3 and 10 mM NaCl, the MNP-content was 20 g/L, the CSA-loading was 0.2, 0.4 and 1.0 mmol -COOH/g MNP and the adsorption time was one day. These amounts of CSA completely coat MNPs and provide the high colloidal stability of the CSA@MNP particles [25].

The HyA-gel was prepared by the dissolution of hyaluronic acid in ultra pure water under one day long continuous stirring. The HyA-content of the stock-gel was 11 mg/mL and the pH was ~5.8. Thereafter, the calculated amounts of the naked MNPs or CSA@MNPs at 0.4 mmol/g CSA-loading were added to 9 mL of the previously prepared HyA-gel under vigorous magnetic stirring and the total volume was adjusted to 10 mL using 10 mM NaCl solutions (pH=6.3±0.3). Stirring was continued for five minutes. The final HyA-content of the hydrogels was 10 mg/mL similar to that in commercial HyA-based intra-articular injections (for example Synocrom). The magnetite-content of the magnetic hydrogels was 2.0 g/L for the naked MNPs and 0.0, 0.2 and 2.0 g/L for CSA@MNPs at pH ~6.0 and ~1 mM NaCl. The HyA-gels were stored in closed vials at 4 °C.

NaCl, HCl and NaOH, analytical grade products of Molar (Hungary), were used to adjust the pH and salt concentration in all experiments. Ultra pure water from a Milli-Q RG water purification system (Millipore) was used. All measurements were performed at 25 ± 0.1 °C.

2.2. Particle Size Determination

For characterization of the aggregation state of magnetic nanoparticles in diluted HyA-systems, the Z average particle hydrodynamic diameter of bare magnetite and of CSA-coated nanoparticles was determined at 25±0.1 °C using dynamic light scattering (DLS) method, an apparatus Nano ZS (Malvern) with a 4 mW He–Ne laser source (λ=633 nm) operating in backscattering mode at an angle of 173°. The dispersions contained 100 mg/L of magnetite to get an optimal intensity of ~10^5 counts per second. Prior to the measurements, the samples were homogenized in an ultrasonic bath for 10 s, after which 2 min relaxation was allowed. For evaluation, second- or third-order cumulant fit of the autocorrelation functions was used, depending on the degree of polydispersity. The influence of the added HyA amounts (0.0–2.5 mmol HyA/g magnetite) was determined at pH=6.3±0.3 and 10 mM NaCl in case of MNPs and CSA@MNPs. The CSA-loadings were 0.2, 0.4, and 1.0 mmol CSA/g magnetite [25].

2.3. Electrokinetic potential measurements

Electrophoretic mobilities of the naked magnetite and CSA@MNP dispersions were measured at 25±0.1 °C in a Nano ZS (Malvern) apparatus using disposable zeta cells (DTS 1060). The zeta-standard of Malvern (-55±5 mV) was used for calibration. The Smoluchowski equation was applied to convert electrophoretic mobilities to electrokinetic potential values. The accuracy of the measurements was ±5 mV. The samples were identical to those in the DLS experiments.
2.4. Rheological measurements

The flow properties of the prepared blank HyA- and magnetic HyA-gels (Section 2.1.) were characterized systematically. The rheological measurements of three-days old samples were carried out with a Physica MCR101 rheometer (Anton Paar, Austria) and a plate-plate measuring device (diameter: 5 cm) at 25±0.1 °C.

Flow curves of the different samples were determined. The shear rate was increased from 0.5 to 50 s⁻¹ (upward curve), and then it was decreased from 50 to 0.5 s⁻¹ (downward curve) in the CR mode. The shearing time in both segments was 60 s. The downward curves over the plastic flow range (above shear rate ~20 s⁻¹) were evaluated according to the Bingham model given by Eq. (1):

\[ \tau = \tau_B + \eta_{pl} \cdot \frac{d\gamma}{dt} \]  

where \( \tau \) is the shear stress, \( \tau_B \) is the Bingham yield value, \( \eta_{pl} \) the plastic viscosity and \( \frac{d\gamma}{dt} \) the shear rate \((42,43))

The creep tests were performed to determine the viscoelastic behavior under static condition. A constant stress (0.1 and 5.0 Pa) was applied for 60 s and the resulting deformation was measured (creep), then stress was released and the deformation was measured again for 60 s (recovery).

The dynamic tests of viscoelasticity were also conducted in forced oscillation measurements in the range of stress from 0.1 to 10 Pa at 1 Hz and of frequency from 0.1 to 10 Hz at 1 Pa. Both storage modulus (\( G' \)) and loss modulus (\( G'' \)) were analyzed.

3. Results and discussion

3.1. Electrokinetic potential, particle size and aggregation

The changes in the Z average particle diameter and the electrokinetic potential of the naked and CSA-coated magnetite nanoparticles are plotted as a function of HyA-addition in Fig. 2. Photos in Fig. 3 show the change in colloidal stability of MNP and CSA@MNP at 0.4 mmol/g CSA-loading as a function of HyA-addition after standing for one day.

![Fig. 2. Effect of HyA-addition on the (a) Z average particle diameter and (b) electrokinetic potential of MNPs and CSA@MNPs (CSA-loadings: 0.2, 0.4 and 1.0 mmol/g) at pH~6.3 and 10 mM NaCl, at a solid/liquid ratio of 100 mg/L. (Lines are drawn as guides for the eye. The amounts of CSA and HyA are expressed as molar amounts of repeating units given in mmol.)](image)
Fig. 3. Effect of increasing HyA addition (up to 2.5 mmol/g, see the numbers on the vials) on the colloidal stability of the naked MNP and CSA@MNP (CSA-loading: 0.4 mmol/g) dispersions after standing for 24 hours at pH~6.3 and 10 mM NaCl, at a solid/liquid ratio of 100 mg/L.

With increasing HyA-concentration, the aggregation state of the originally naked MNPs changes characteristically (see Fig. 3). At pH~6.3 and 10 mM NaCl, the naked magnetite is aggregated and settled [23,24]. Although the net surface charge of MNP is positive due to the presence of ≡Fe–OH₂⁺ groups on the surface [23,24,32,33,37,44-46], the amount of the charge is not enough to stabilize the particles electrostatically. The electrokinetic potential of the naked magnetite is ~+25 mV, and this sample is aggregated (~1000 nm) (see Fig. 2). In case of trace (<0.5 mmol/g) or very large (>1.5 mmol/g) amounts of HyA, the particles settle after 24 hours. However, at intermediate HyA-concentrations (~0.5-1.5 mmol/g) the magnetite dispersions are stable (middle samples in Fig. 3). The electrokinetic potential of the positive naked MNPs declines with the increase in the amount of added HyA because the negatively charged HyA adsorbed on MNPs. The charge neutralization occurs at ~0.05 mmol/g HyA-addition, resulting in surface charge heterogeneity due to the patch-wise adsorption of HyA. This marks the isoelectric point (IEP) and the size of HyA@MNP aggregates reaches maximum (~2300 nm). Further addition of HyA caused charge reversal and reduced the electrokinetic potential to ~40 mV. The electrosteric stabilization, typical for polyelectrolytes, can be observed only in the range of ~0.5-1.5 mmol/g HyA. Stabilization is accompanied by a low hydrodynamic diameter (~250 nm). The aggregation observed at larger HyA-addition (>1.5 mmol/g) is caused by the flocculation effect of the high molecular weight HyA, so these particles do not seem to be suitable for preparation of magnetic HyA-gels.

The colloidal state of the originally stable CSA@MNPs [25] does not change with increasing HyA concentration (Fig. 3). The hydrodynamic diameters (~150 nm) and the electrokinetic potentials (~40 mV) are independent from the CSA-loadings and the HyA-addition (Fig. 2). Thus, the CSA@MNP particles can be used to prepare magnetic HyA-gels.

3.2. Rheological measurements of hydrogels

The photos of the HyA-hydrogels prepared at 10 mg/mL HyA concentration are shown in Fig. 4. As it was expectable from the DLS measurements of the diluted gels (Section 3.1.), the naked MNPs can not be dispersed homogeneously in the HyA-gel; however, the CSA@MNP particles are compatible with the hydrogels forming well dispersed systems. The CSA@MNP-containing magnetic hydrogels remained stable and homogeneously dispersed even after three months of storage.
Fig. 4. HyA-gels with MNP and CSA@MNP at different magnetite contents (see on the vials) (HyA-content: 10 mg/mL, CSA-loading: 0.4 mmol/g, pH~6.0, 1 mM NaCl).

The flow curves of the HyA-hydrogels at various CSA@MNP-contents are shown in Fig. 5. There is no significant change in the flow character of gels with increasing CSA@MNP-loading. The upward and downward curves are identical. Thixotropy is not detected under the conditions of measurements. The downward curves were evaluated according to the Bingham-model (Fig. 5). The Bingham yield values ($\tau_B$) and the plastic viscosities ($\eta_p$) of all gels (Table 1) coincide within the error of the measurements. This is in good agreement with the expected tendency based on the Einstein-equation [47], since the volume fraction of magnetite particles is extremely low in these hydrogels ($\Phi \leq 0.004$).

Fig. 5. Effect of CSA@MNP-content on flow curves of HyA-gels (HyA-content: 10 mg/mL, CSA-loading: 0.4 mmol/g, pH~6.0, 1 mM NaCl, 25 °C). (The upward and downward curves are identical. The equations of fitted Bingham model are marked.)

The viscoelastic properties of gels can be studied under static conditions by creep tests. The measured data of the HyA-gels at different CSA@MNP contents are shown in Fig. 6 at two yield stress values (0.1 and 5.0 Pa). Both the HyA-gel and the magnetic hydrogels show viscous properties at the applied stress values showing that the extent of deformation depends on the applied stress, but it is independent of the magnetite content.
The forced oscillation measurements are suitable to study the viscoelastic properties of gels under dynamic conditions. The oscillation tests of the HyA-gels at different CSA@MNP-contents (Fig. 7) show that all the measured parameters are independent from the CSA@MNP-content within the error of the measurements.

During the oscillation measurements at 1 Hz (Fig. 7a) the loss modulus (G'') curves of gels are above the storage modulus (G') curves in the whole range of stress (0.1 to 100 Pa). Thus, under the condition of measurements the hydrogels show viscous properties. However, during frequency sweep measurements at 1 Pa (Fig. 7b), the viscous loss modulus (G'') and the elastic storage modulus (G') of gels cross each other. This is characteristic for the viscoelastic nature of HyA, i.e., at lower frequencies (analogous to the static position of the body) the HyA-gel behaves as a viscous solution ensuring to hold the skeleton, however, at higher frequencies (analogous to the motion of the joints) the hydrogel behaves elastically allowing an effective lubrication between the cartilages \[48-50\]. This property is fundamental and essential for the natural synovial fluid. The transition from viscous to elastic behaviour can be characterized by the so-called crossover point (Table 1), which is at \(\sim 1.3\) Hz for our magnetic HyA-gels in case of 1 Pa shear stress.

**Fig. 6.** Creep and recovery test results for HyA-gels at different CSA@MNP-content (HyA-content: 10 mg/mL, CSA-loading: 0.4 mmol/g, pH~6.0, 1 mM NaCl, 25 °C).

**Fig. 7.** Forced oscillation tests of HyA-gels at different CSA@MNP-content as a function (a) of shear stress at 1 Hz and (b) of frequency at 1 Pa. (HyA-content: 10 mg/mL, CSA-loading: 0.4 mmol/g, pH~6.0, 1 mM NaCl, 25 °C).
The characteristic rheological properties of the magnetic HyA-gel loaded with 2 g/L CSA@MNP (indicated as HyA + magnetite–COO⁻) can be compared with those of other HyA-based magnetic hydrogels as shown in Table 2. It can be seen that the values of the different parameters for the listed hydrogels vary significantly, probably because these measurements were made under different conditions, such as temperature, salt concentration, pH, fixed frequency or stress, etc. There were differences even in the rate of cross-linking. Our magnetic hydrogels are not yet cross-linked, however, the un-bound carboxylate groups of both CSA and Hyal (as seen in Fig.1) are prone to react with Ca²⁺ ions forming a cross-linked structure with tuneable flow properties. The latter will be the subject of our future studies.

### Table 2. Characteristic rheological parameters of different HyA-hydrogels.

<table>
<thead>
<tr>
<th>Hydrogels</th>
<th>Magnetic</th>
<th>Cross-linked</th>
<th>at linear range of stress sweep</th>
<th>at crossover point in frequency sweep</th>
<th>from flow curve</th>
</tr>
</thead>
<tbody>
<tr>
<td></td>
<td></td>
<td></td>
<td>G' (Pa)</td>
<td>G'' (Pa)</td>
<td>τ_{pl} (Pa)</td>
</tr>
<tr>
<td></td>
<td></td>
<td></td>
<td>f (Hz)</td>
<td>G'=G'' (Pa)</td>
<td>η_{pl} (Pa·s)</td>
</tr>
<tr>
<td>HyA + magnetite–COO⁻¹</td>
<td>yes</td>
<td>no</td>
<td>11</td>
<td>13</td>
<td>0.2</td>
</tr>
<tr>
<td>HyA + magnetite–NH₂ [15]²</td>
<td>yes</td>
<td>yes</td>
<td>350</td>
<td>25</td>
<td>0.1</td>
</tr>
<tr>
<td>HyA-SH + iron-oxide–COOH [16]³</td>
<td>yes</td>
<td>yes</td>
<td>510</td>
<td>-</td>
<td>-</td>
</tr>
<tr>
<td>HyA [50]</td>
<td>no</td>
<td>no</td>
<td>-</td>
<td>-</td>
<td>-</td>
</tr>
<tr>
<td>HyA [50]</td>
<td>no</td>
<td>yes</td>
<td>-</td>
<td>-</td>
<td>-</td>
</tr>
<tr>
<td>HyA – Synvisc [51] ⁴</td>
<td>no</td>
<td>yes</td>
<td>100</td>
<td>20</td>
<td>0.1</td>
</tr>
</tbody>
</table>

¹ magnetite: chondroitin-sulfate-A-coated

² magnetite: 3-aminopropyltrimethoxysilane-functionalized

³ HyA: modified with hydrazide and thiol groups, iron-oxide: oleic acid coated in toluene

⁴ commercially available intra-articular injection

### 4. Conclusions

The aim of this work was to prepare novel biocompatible HyA-based magnetic hydrogels using CSA-coated superparamagnetic magnetite nanoparticles and to characterize their aggregation state and rheological properties as well. The solid experimental facts allow us to conclude that HyA-based magnetic hydrogels can be prepared from CSA@MNP and they remained stable and homogeneously dispersed even after three months of storage. Rheological studies show, that at pH~6.0 and 1 mM NaCl the behaviour of magnetic hydrogels was appreciably independent from the magnetite content up to 2 g/L. Further studies are needed at pH ~7.3, 150 mM NaCl and 37 °C to approximate the composition of commercial HyA-based intra-articular injections and in vivo conditions. However, the prepared HyA-based magnetic hydrogels are promising candidates for magnetic intra-articular fluid in the treatment of osteoarthritis. By implanting a permanent magnet into the bone in close vicinity of the knee joint for example, the magnetic HyA-gel with increased CSA@MNP content probably can be held at the eroded area. Thus, it is predictable that the necessary distance between the cartilages

---

**Table 1.** Summary of the hydrogel’s characteristic rheological parameters at different CSA@MNP contents

<table>
<thead>
<tr>
<th>CSA@MNP magnetite content</th>
<th>0.0 g/L</th>
<th>0.2 g/L</th>
<th>2.0 g/L</th>
</tr>
</thead>
<tbody>
<tr>
<td>Bingham yield stress (Pa)</td>
<td>28.0</td>
<td>27.1</td>
<td>30.5</td>
</tr>
<tr>
<td>plastic viscosity (Pa·s)</td>
<td>0.42</td>
<td>0.42</td>
<td>0.46</td>
</tr>
<tr>
<td>crossover point (Hz) ¹</td>
<td>1.3</td>
<td>1.1</td>
<td>1.4</td>
</tr>
</tbody>
</table>

¹ intersection point between the viscous and the elastic behaviour in forced oscillation during the frequency sweep at 1 Pa.
can be provided to reduce the pain during motion and even the requested lubrication can be ensured in the joint for a longer time.

Acknowledgment
This research was supported by the OTKA (NK 84014) grant as well as by European Union and the State of Hungary, co-financed in the framework of TÁMOP 4.2.4. A/2-11-1-2012-0001 ‘National Excellence Program’. We thank Erzsébet Csányi (University of Szeged, Department of Pharmaceutical Technology) for rheology measurements.

References


