



Graphitic Nature Governs CO₂ Hydrogenation Reactions on Platinum@Carbon Nanocomposites

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Abstract

In this study, we discuss the influence of carbon support on the performance of platinum nanoparticles (Pt-NPs) in gas-phase CO₂ hydrogenation towards industrially demanding CO production. To this end, Pt-supported on graphite oxide, carboHIPE, activated carbon, multiwalled carbon nanotubes (MWCNT), and graphite of different specific surface areas and degree of graphitization were synthesized with identical loading per unit surface area (25 µg Pt/m² of support) to reveal the contribution of the carbon support. All Pt@carbon nanocomposites were selective to CO at temperatures above 800 K, however, we found an additional correlation between their catalytic performance and the degree of graphitization of the support. Graphite-supported platinum (Pt@Graphite) has the highest TOF and the lowest activation energy among the studied nanocomposites due to the increasing electron donation from the carbon support to the Pt nanoparticles, which facilitates CO₂ adsorption on Pt by shifting its d-band centre towards the Fermi level. Our study shows that beside surface area, porosity, thermal stability, a further parameter needs to be taken into consideration, as the Pt/C interface has significant effect at the atomic level on the activity of carbon-supported catalysts in gas-phase CO₂ hydrogenation towards CO production.

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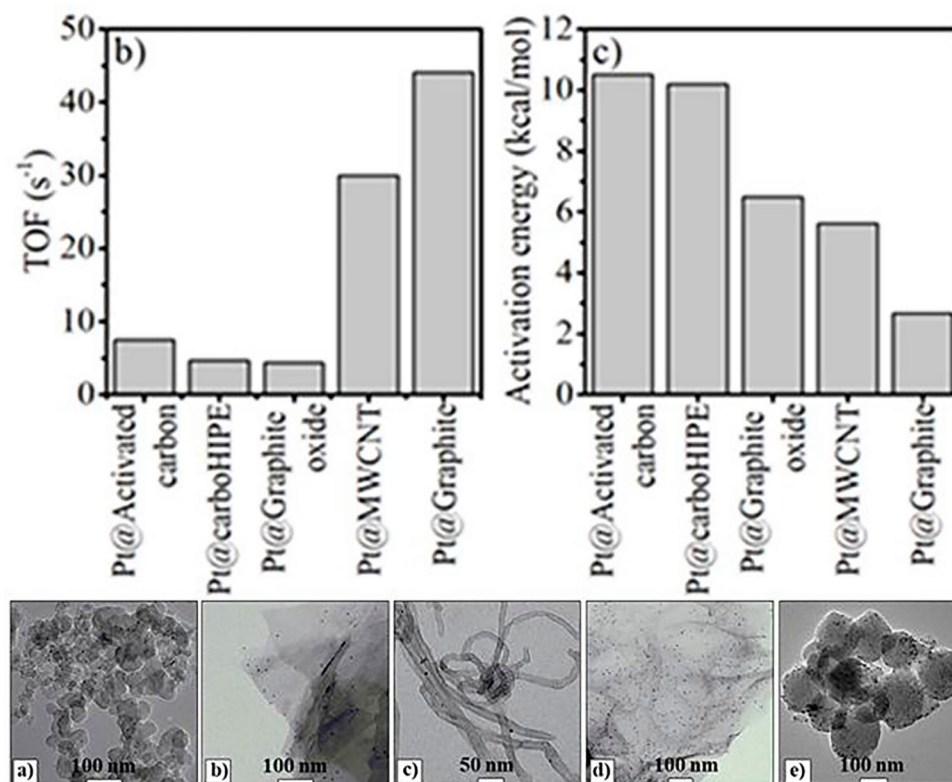
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Graphical Abstract



Keywords CO₂ hydrogenation · Carbonaceous supports · Graphitic nature · Pt nanoparticles · RWGS

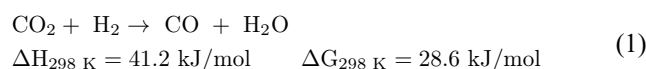
1 Introduction

The development of a sustainable economy requires a balance between the ecological, social, and economic goals within social and industrial processes. In terms of the IPCC (Intergovernmental Panel on Climate Change) [1], Carbon Dioxide Removal alternatives are needed to compensate for residual emissions from a likely climate overshoot of the targets set in the Paris Agreement. At the industrial level, there is a portfolio of mitigation alternatives with varying levels of development and feasibility. They can be divided into three strategies: reduced GHG (greenhouse gas) emissions, CO₂ Capture and Storage (CCS), and CO₂ utilization [2]. CO₂ utilization processes chemically alter the CO₂ molecule, unlike other technologies that aim to reduce overall CO₂ emissions while leaving the molecule unchanged [3]. CO₂ utilization processes have emerged as a potential complement to CCS and have a promising competitive advantage for a more sustainable industry. A variety of products (e.g., fuels, chemicals or materials) can be synthesized via CO₂ utilization by the appropriate technologies [4]. One such example is the so-called Power-to-Gas (PtG) technology, which uses H₂ and CO₂ to produce alternatives to

natural gas [5]. However, this technology still faces many challenges, such as sustainable H₂ production, efficient CO₂ capture from industrial waste streams, and scaling. Nevertheless, these low- and negative-emission technologies can provide an alternative to reduce the exploitation of fossil energy sources and reduce the environmental impacts associated with energy consumption [6].

A CO₂ recycling process requires catalytic reduction into various products, mainly methane, carbon monoxide, and methanol. Afterwards, these products can be further converted into other chemicals through additional steps such as the Fischer–Tropsch process [7]. Typically, the optimization of the CO₂ reduction process is a balance between the endothermic Reverse Water Gas Shift Reaction (RWGS) and the thermodynamically favorable CH₄ formation by the Sabatier process. The key reactions for these processes are listed below:

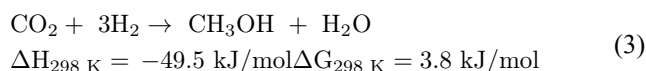
Reverse water gas shift reaction (RWGS)



Methane formation



Methanol formation



Pt, Cu, and Rh have been identified as the most efficient catalysts for the RWGS reaction, and several key parameters have been found to enhance the catalytic performance. A recently demonstrated example is the synergistic effect between Pt nanoparticles and different substrates (SiO₂ and TiO₂) [8], and another example is the probable correlation between the catalytic activity of industrial catalysts (Cu/ZnO/Al₂O₃) for methanol synthesis and the density of defects and twin boundaries [9, 10]. However, explaining catalytic performance still poses a challenge, as for the detailed mechanistic understanding morphology must be taken into account besides the thermodynamic processes involved at given reaction conditions. For example, the reversible migration of zinc atoms towards the copper surface at high temperatures, the partial reduction by the hydrogen reactant, and the role of supports and defects have been reported in the literature [10, 11]. In recent years, various metal@carbon nanocomposites, for instance Pt [12–17], Fe [18], Ni–ZrO₂ [19] or Ni–SiO₂ [20] nanoparticles to name a few, have been investigated for electrocatalytic applications, CO₂ hydrogenation or Fischer–Tropsch catalysis.

To investigate the influence of carbon supports in the gas-phase CO₂ hydrogenation reaction, we prepared and studied Pt@carbon nanocomposites based on different carbonaceous matrices, such as carbon nanotubes, graphite, activated carbon, graphite oxide, and carboHIPE. These supports differ mostly by the degree of graphitization and the specific surface area. The loading of the controlled size Pt-NPs was set to 25 µg/m² of carbon support, and significant differences were found in the catalytic activity between samples. Turnover frequencies (TOF) and activation energies reveal the strong influence of the carbonaceous matrix on the performances of the Pt@carbon nanocomposites. It was found, that higher carbon graphitization degree leads to higher TOF values, whereas the overall CO₂ consumption rate strongly depends on the specific surface area.

2 Experimental Section

2.1 Carbon Materials

Activated carbon was purchased from Sigma-Aldrich and used as-received without any further modification.

CarboHIPE support was produced as described elsewhere [21]. Briefly, for the preparation of carboHIPE, first, the poly-HIPE precursor was synthesized through oil-in-water (O/W) high internal phase emulsion (HIPE) consisting of aqueous solution as the minority phase and toluene as the majority phase. The aqueous solution was composed of acrylamide (AAM, Sigma-Aldrich), N,N'-methylene(bisacrylamide) (MBAM, Sigma-Aldrich), surfactant (Pluronic F-127), ammonium persulfate (APS, Fluka), and N,N,N',N'-tetramethylethylenediamine (TEMED, Sigma-Aldrich). Once the AAM-based polyHIPE was obtained, the carbonization of the purified polyHIPE was performed in a tube furnace, reaching 900 °C at a heating rate of 5 °C min⁻¹.

For the synthesis of graphite-oxide, 4.5 g graphite powder (Sigma-Aldrich) was sonicated in 210 ml cc. Sulphuric acid (Molar) for 20 min. After sonication and during vigorous mixing, 4.5 g NaNO₃ was added to the suspension. After 30 min the suspension was placed on an ice bath, while 27 g potassium-permanganate was slowly added to the slurry. After 24 h of mixing, first 500 ml of distilled water, then 10 ml 50% H₂O₂ was added to the mixture. After 2 h of mixing, the product was washed and collected by centrifugation.

Multiwall carbon nanotubes were produced by CCVD using acetylene precursor and alumina supported 2.5–2.5% iron-cobalt bimetallic catalyst. The system was heated to 700 °C under nitrogen atmosphere in a tube reactor, then C₂H₂:N₂ mixture of 1:9 ratio with a total flow rate of 300 ml min⁻¹ was introduced for 30 min. After cooling down in pure N₂ atmosphere, the as-obtained material was suspended in NaOH solution to remove the alumina support, then washed with distilled water. The metal catalyst particles formed at the beginning of the synthesis procedure were dissolved in diluted nitric acid solution. The amorphous carbon was removed by potassium permanganate in acidic media. Carbon nanotubes were finally washed with distilled water to neutral pH and dried in air.

2.2 Synthesis of Size-Controlled Pt Nanoparticles

50 mg H₂PtCl₆ and 220 mg polyvinylpyrrolidone (PVP, MW=40.000) were dissolved in 10 mL ethylene-glycol and sonicated for 30 min to get a homogeneous solution [22]. The flask was evacuated and purged with atmospheric pressure argon gas. After three purging cycles, the flask was immersed into an oil bath and heated to 433 K for 60 min under vigorous stirring. After reaction, the flask was cooled down to room temperature and sonicated with acetone. After precipitation the particles were washed by centrifugation (6000 RPM, 10 min) with a mixture of 2 mL ethanol and 8 mL hexane. Finally, the particles were re-dispersed in ethanol.

2.3 Catalysts Preparation

To unveil the effect of the carbonaceous support, nanocomposite catalysts with identical platinum (Pt)-nanoparticles per unit surface area loading needed to be prepared. For the preparation of Pt-carbon (Pt@C) nanocomposites, 500 mg of each carbon support was sonicated in 40 mL ethanol to obtain a suspension. A precise volume of platinum was added to the suspension of carbon material from a 1500 ppm ethanol suspension of size-controlled Pt nanoparticles (4–5 nm) to reach $25 \mu\text{g m}^{-2}$ Pt loading.

2.4 Catalytic Activity Tests

For the catalytic experiment, a 60 min long pretreatment was applied at 573 K with 40 mL min^{-1} of hydrogen flow for all samples. After the pretreatment, the catalyst was cooled down to 473 K and allowed to settle for 5 min in continuous N_2 flow. The gas source was switched to a premixed $\text{CO}_2:\text{H}_2$ flow with the constant molar ratio of 1:4 ($10:40 \text{ mL min}^{-1}$). The reaction started at 473 K, then the system was heated up to 873 K in 160 min. The CO_2 conversion to CO and CH_4 was detected at temperatures higher than 673 K.

2.5 Characterization

The as-prepared catalysts were characterized by Transmission Electron Microscopy (TEM, FEI Tecnai G² 20 X-TWIN operated at 200 kV). Raman spectra were collected with a Bruker Senterra II Raman microscope equipped with a 532 nm excitation laser in backscattering configuration. The crystalline structure of the materials was examined by X-ray Diffraction (XRD, Rigaku MiniFlex II Desktop Diffractometer with a $\text{CuK}\alpha$ source ($\lambda = 0.1542 \text{ nm}$) operated at 30 kV and 15 mA). Surface areas were determined by nitrogen sorption measurements using a Quantachrome NOVA 3000e instrument and calculated using the Brunauer–Emmett–Teller (BET) model. Thermogravimetric analysis was conducted in a TA Instruments, Q500 instrument using platinum sample holder under oxidative and inert atmospheres (synthetic air and nitrogen, respectively) at a heating rate of 5 K min^{-1} .

The Pt loading of the obtained catalysts was determined by inductively coupled plasma mass spectrometer (ICP-MS) using an Agilent 7900 instrument. 5 mg of each sample was digested in 5 ml high-purity aqua regia (made of ICP-MS grade cc. HNO_3 and cc. HCl) for 4 h at 75°C . The samples were shaken several times during digestion to aid platinum dissolution. After cooling, 4 ml of the digested mixture was filtered using $0.45 \mu\text{m}$ PES membrane filters and was brought to volume in a 100 ml volumetric flask with deionized water. The platinum concentration of the

appropriately diluted aliquots was measured, where each aliquot contained 0.1 ppm mixture of rare earth elements (Aristar, VWR Chemicals) used as internal standards. The ^{194}Pt , ^{195}Pt , and ^{196}Pt isotopes were used for quantification, based on a 11-point calibration series in the 0–100 ppb range (Aristar, VWR Chemicals). Ion counts were measured both with and without a helium collision cell.

XPS spectra were recorded in a Kratos XSAM 800 X-ray photoelectron spectrometer. Double sided carbon tape was used to stick 50 mg sample to a stainless-steel sample holder. The Al $\text{K}\alpha$ x-ray source was operated at 120 W (12 kV). During measurement no sample charging was observed. Survey and high-resolution spectra (C 1 s, O 1 s, and Pt 4f) were collected with a pass energy of 80 and 40 eV, and a step size of 1 and 0.1 eV, respectively. The data was evaluated in the CasaXPS software package [23], where high resolution spectra were corrected with a Shirley background and the aliphatic component of the C 1 s spectra were used as an inner reference at 284.8 eV binding energy. Peaks characteristics of metallic Pt were fit with an asymmetric Lorentzian peak shape (LA(1.2,85,70)).

3 Results and Discussion

3.1 Characterization of Pt@C Nanocomposites

Various carbon matrices, namely graphite oxide, carbo-HIPE, activated carbon, MWCNT, and graphite were used as supports for the Pt-NPs, and the microstructure of the Pt@carbon nanocomposites was characterized by electron microscopy. The TEM images of the nanocomposites in Fig. 1a–e reveal a uniform distribution of $5.0 \pm 0.6 \text{ nm}$ Pt-NPs (Fig. 2) without any apparent particle aggregation.

Since the catalytic activity of the Pt@carbon nanocomposites was tested as a function of different carbon supports, carbon matrices were characterized by X-ray diffraction (XRD), Raman spectroscopy and N_2 -physisorption in detail before any CO_2 hydrogenation reaction. Figure 2 shows the X-ray diffraction (XRD) pattern of the different carbon supports. The diffraction peaks around 26° and 43° 2-theta (2θ) indicate the crystalline nature of the graphitic structure for the graphite and graphite oxide supports. The other carbon supports show only one broad reflection at about 23° (2θ), which is likely due to their amorphous structure.

Since catalytic CO_2 hydrogenation was carried out at temperatures above 673 K, the thermal stability of the carbon matrix is of paramount importance. Thus, thermal gravimetric analysis (Figure S1) and high-temperature Raman spectroscopy (Figures S2–6) were carried out. Thermal stability in an oxidative atmosphere (synthetic air) at temperatures up to 873 K showed that MWCNTs and graphite are the

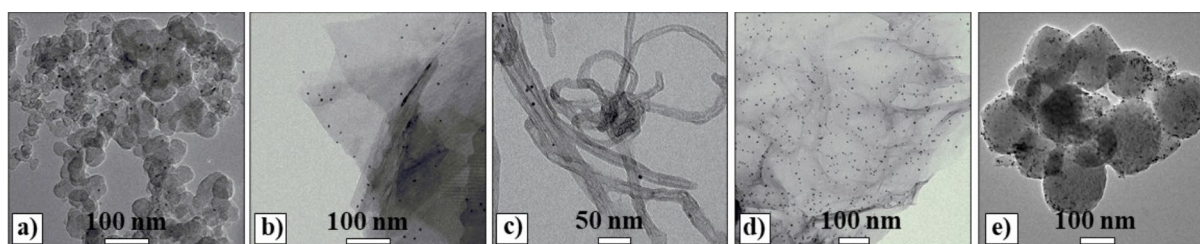


Fig. 1 Platinum nanoparticles on **a** activated carbon, **b** graphite, **c** MWCNT, **d** graphite oxide, **e** carboHIPE

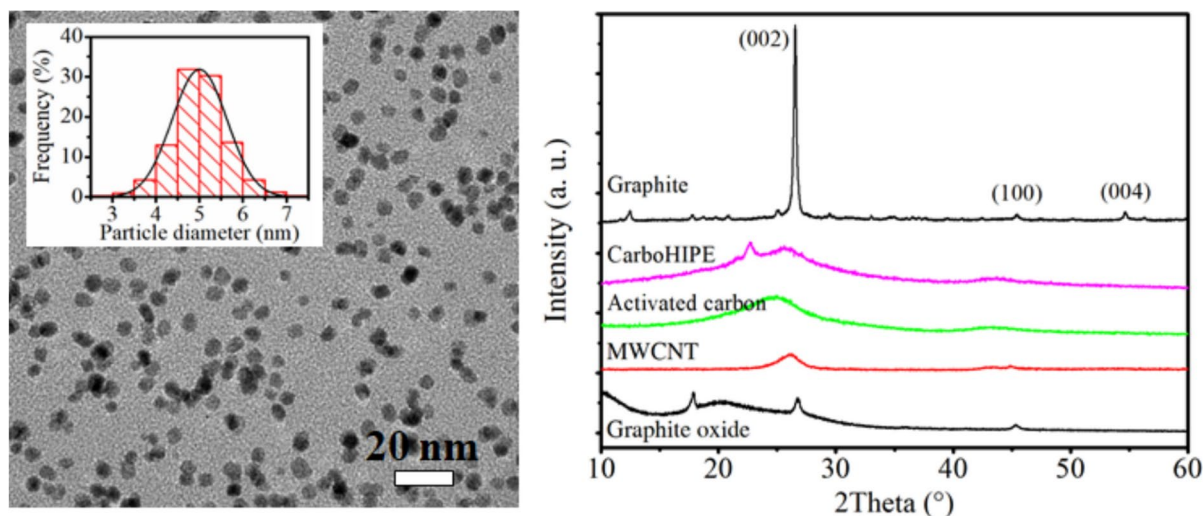


Fig. 2 Size-controlled platinum nanoparticles with a diameter of 5.0 ± 0.6 nm, along with the corresponding particle size distribution in the inset (left), and XRD pattern of the carbon supports (right)

most stable carbon supports. Their decomposition started at about 823 K in good agreement with previous reports [24]. Activated carbon proved to be a moderately stable support, remaining stable up to temperatures above 773 K. Finally, the lowest thermal stability was found for graphite oxide and carboHIPE, where decomposition occurred at around 673 and 700 K, respectively (Figure S1). The high thermal stability of MWCNTs was confirmed by Raman spectroscopy, revealing thermal expansion without any sign of degradation (Figure S4). On the other hand, even though TGA indicated the superior thermal stability of graphite among carbon matrices, Raman spectra show signs of degradation. The intensity of the otherwise weak vibrational band at 643 cm^{-1} increases with temperature above 523 K, which is likely to be related to more pronounced C-H bending deformations (Figure S3). In Pt@activated carbon and Pt@graphite oxide a monotonic increase in I_D/I_G ratio is seen as a sign of degradation. Activated carbon (Figure S2) and graphite oxide (Figure S5) have similar Raman spectra, as both have D and G bands located at 1345 and 1585 cm^{-1} , respectively. In addition, graphite and MWCNTs have similar Raman spectra with an intense peak around 1580 cm^{-1} (G), which is the first-order high-frequency E_{2g} mode. Furthermore, these two carbon matrices show lines corresponding to

second-order frequency modes of D', G', D+G, and 2D vibrational bands, and have not been found in other samples (Figure S3 and S4). The lowest signal-to-noise ratio was found for the carboHIPE sample as this carbon matrix emits intense fluorescence during Raman measurements, even when 785 nm laser excitation was used. Nevertheless, Figure S6 confirms the presence of D and G bands at around 1328 cm^{-1} and 1590 cm^{-1} , respectively.

Raman spectroscopy is known as a versatile technique to study defects and the degree of graphitization in carbon materials, and can be used to determine the number of defect sites in carbon supports [25]. The G-band is attributed to sp^2 hybridized carbon, while the D-band is associated with disordered carbon. It is known that the intensity of the D-band increases as the number of defects in the carbon matrix increases. Thus, we have calculated the intensity ratio of the two bands (I_D/I_G) and use this value as an indicator of the change in defect site density. I_D/I_G ratios of 0.65, 1.24, 5.09, and 9.42 were found for graphite, MWCNT, graphite oxide, and activated carbon, respectively. No exact I_D/I_G ratio could be obtained for carboHIPE due to the low signal-to-noise ratio of its spectrum. Hence, the defect site density increases in the following order: graphite < MWCNT < Graphite oxide < Activated carbon.

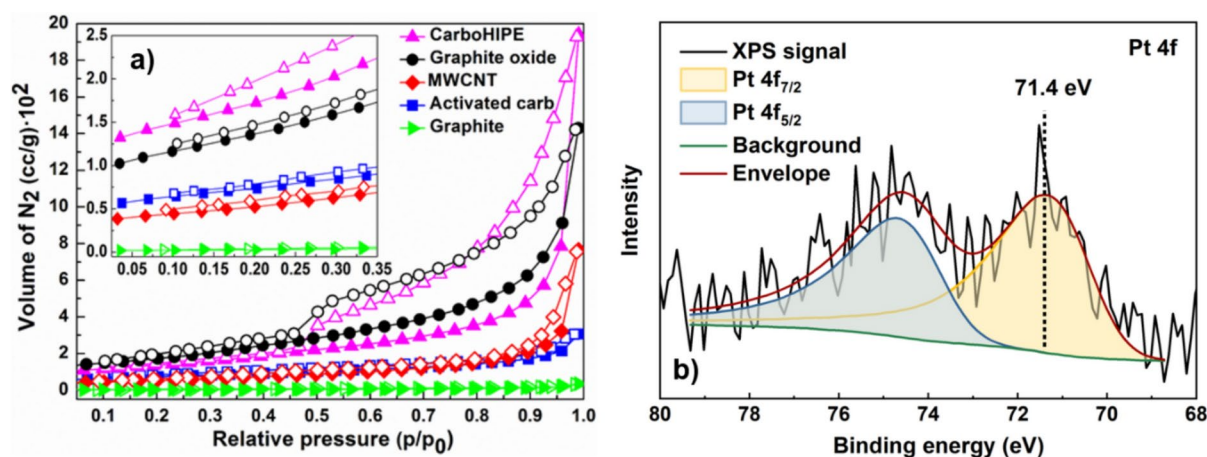


Fig. 3 N_2 sorption isotherms measured at 77 K. The inset shows N_2 uptake in the low relative pressure range ($P/P_0 < 0.35$) (a), and the high-resolution XPS spectrum of Pt 4f of Pt/carboHIPE fitted with asymmetric peak shapes for elemental platinum (b)

Table 1 Specific surface area (SSA), I_D/I_G ratio characterizing defect density, crystallite size from Eq. 4 and the Scherrer-equation, catalytic performance (TOF at 873 K, CO_2 consumption rate), and the Specific Degree of Graphitization (SGD) index defined in this study

Sample	SSA (m^2/g)	I_D/I_G	La (nm)	Crystallite size (nm)	Pt (ICP-MS) (w/w %)	TOF _{873 K} (s^{-1})	Consumption rate ($nmol/m^2/s$)	SGD Index (nm/m^2)
Pt@Activated carbon	255	9.42	2.04	1.22	0.68	7.45	35.16	0.0096
Pt@CarboHIPE	266	—	—	0.78	0.69	4.67	13.10	0.0059
Pt@Graphite oxide	624	5.09	3.77	1.36	1.57	4.35	20.53	0.0044
Pt@MWCNT	194	1.24	4.10	3.81	0.51	30.00	87.64	0.0393
Pt@Graphite	11	0.65	29.69	31.42	0.03	44.04	206.61	5.7130

In heterogeneous catalysis the use of porous supports is favorable, as the catalytically active metal centers can be distributed over a larger surface area, which, in turn, leads to higher catalytic performances. Porosity of the Pt@carbon nanocomposites was characterized by the specific surface areas (S_{BET}) calculated from the corresponding nitrogen adsorption/desorption isotherms. Results in Fig. 3 show typical IUPAC type-IV adsorption/desorption behavior with pronounced H3 hysteresis loops. The latter confirms the broad pore size distribution over the micro-, meso- and macropore ranges [26, 27]. The steep increase in N_2 uptake at low relative pressures ($P/P_0 < 0.1$) in carboHIPE, graphite oxide, MWCNT, and activated carbon (Fig. 3 inset) and the hysteresis loops at higher pressures ($0.5 < P/P_0 < 0.8$) in graphite oxide, carboHIPE, and to some extent in MWCNT, are features of micro- and meso-porous structures, respectively. The steep increase at $P/P_0 \approx 1$, on the other hand, indicates the presence of macropores in carboHIPE. The specific surface area was calculated using the BET method, and values for S_{BET} were found to be 624, 266, 255, 194, and $11 \text{ m}^2 \text{ g}^{-1}$ for graphite oxide, carboHIPE, activated carbon, MWCNT, and graphite, respectively. Graphite showed a significantly lower S_{BET} of $11 \text{ m}^2 \text{ g}^{-1}$ than the other carbon matrices, which can be attributed to the negligible micro-pore volume. This is further confirmed by the almost zero N_2 uptake at $P/P_0 < 0.1$ (Fig. 3 inset).

Based on the foregoing S_{BET} values of the carbon supports of various graphitization, size-controlled 4–5 nm Pt nanoparticles were deposited to reach $25 \mu\text{g m}^{-2}$ Pt loading, i.e., 1.536, 0.661, 0.633, 0.483, and 0.027 wt% platinum on carbon for graphite oxide, carboHIPE, activated carbon, MWCNT, and graphite, respectively. The actual Pt loadings were determined by ICP-MS and loadings of 1.568, 0.692, 0.678, 0.509, and 0.032 wt% were obtained on GO, carboHIPE, AC, MWCNT, and graphite-supported catalysts, respectively (Table 1).

The Pt 4f XPS spectrum of the Pt/carboHIPE sample is plotted in Fig. 3b. The binding energies of 71.4 eV and 74.7 eV of the single platinum species are in good agreement with those found for $4f_{7/2}$ and $4f_{5/2}$ of the Pt^0 state in bulk platinum [28]. Oxidized Pt species have slightly higher binding energies of around 75 and 78 eV for $4f_{7/2}$ and $4f_{5/2}$ of $Pt(IV)$ in PtO_2 or $Pt(OH)_4$ [29]. The valence of platinum in Pt NPs depend, among others, on the synthesis route, where by varying the precursor and the reduction process, Pt NPs of different $Pt(0)/Pt(IV)$ [30] or even $Pt(0)/Pt(II)/Pt(IV)$ ratios [31] were obtained. Our polyol synthesis route, on the other hand, provides highly metallic Pt NPs, as we already pointed out by depositing Pt NPs onto hexagonal boron-nitride and SBA-15 supports. Here, no significant charge transfer occurs between the platinum nanoparticles and the support, and hence, the inherent highly metallic state

of the particles was seen [32]. Supports can also modify the initial valence of a metal nanoparticle, and Pt NPs are more metallic on more electrophilic supports or using more electrophilic additives [33]. Studies showed that platinum on acidic supports is less oxidized than on basic ones under an oxidizing atmosphere, whereas in a reducing environment Pt is more electron-deficient on acidic supports [34]. Earlier, we deposited our Pt NPs onto NiO, where Pt got partially oxidized, while operando NAP-XPS measurements showed that during CO₂ hydrogenation at 673 K the oxidized Pt fraction decreased. After the reaction the Pt(δ^+) content regained due to oxygen transfer between NiO and the Pt [35]. Although surface oxygen can be found on the Pt (111) surface under pure CO₂ atmosphere at room temperature by ambient pressure XPS, it was removed by the presence of hydrogen and the increase in temperature under reaction conditions. Above 500 °C RWGS reaction prevailed, and surface -OH and H₂O were solely found on the platinum surface. In our study, we performed a reduction step in flowing hydrogen at 573 K for 60 min (see ‘Experimental Section’) before every test run, and under the reaction conditions used throughout our study, the presence of surface oxygen is highly unlikely. Furthermore, carbon supports possess poor activity in acid-catalyzed reactions, and can be made active only by, e.g., sulfonation or similar chemical process [36]. Since our carbonaceous materials were not activated but used as-prepared, no solid acid-related oxidation state change is expected.

3.2 Hydrogenation Reaction Over Different Pt@C Based Nanocomposites

All Pt@carbon nanocomposite catalysts were activated *in-situ* in a pure H₂ feed at 573 K before being tested in the CO₂ hydrogenation reaction at up to 873 K. The catalytic performances of the Pt@carbon nanocomposites were evaluated based on their selectivity, CO₂ consumption and conversion rate, activation energy, and TOF. The reaction conditions were adjusted so that we achieved relatively low

conversion rates, i.e., far from the thermodynamic equilibrium. This ensures that the reported rates correspond to the intrinsic specific activity of the metal catalysts. The conversion rate is expressed as nanomoles of carbon consumed per gram of catalyst per second ($\text{nmol} \cdot \text{g}^{-1} \cdot \text{s}^{-1}$).

Figure 4 plots the CO₂ consumption rate at different temperatures (Fig. 4a) and the isotherms at 873 K (Fig. 4b) for CO₂/H₂ (1:4 ratio). An increase in catalyst activity with increasing reaction temperature is observed for all Pt@carbon nanocomposites, reaching its maximum at 873 K with CO₂ consumption rates of 17,000, 12,813, 8,965, 3,415, and 2,272 $\text{nmol} \cdot \text{g}^{-1} \cdot \text{s}^{-1}$ for Pt@MWCNT, Pt@Graphite oxide, Pt@activated carbon, Pt@carboHIPE, and Pt@graphite, respectively. However, all catalysts showed relatively low catalytic activity at temperatures below 673 K, as their corresponding CO₂ consumption rates were below 1200 $\text{nmol} \cdot \text{g}^{-1} \cdot \text{s}^{-1}$ (Fig. 4a). Interestingly, the CO₂ consumption rates of Pt@graphite and Pt@carboHIPE are reasonably similar (Fig. 4b), although the latter has a significantly larger S_{BET} compared to that of the former. It seems that other variables and the poor thermal stability of carboHIPE are crucial factors for the low CO₂ consumption rate at 873 K compared to the other carbon matrices with similarly high S_{BET} .

Another important observation is, that Pt@carbon nanocomposites show different selectivity for CO and CH₄ (Fig. 5), indicating an important role of the carbon support in the adsorption and dissociation of CO₂. Pt@Graphite oxide and Pt@MWCNT nanocomposites showed 100% selectivity for CH₄ (methanation reaction), despite the lower CO₂ consumption rates at 630 K. With an increasing temperature CH₄ selectivity starts to decrease and levels off at almost 0% above 750 K. Surprisingly, in parallel to that, both samples reach 100% selectivity for CO (RWGS reaction) at temperatures above 750 K (Fig. 5a, b). A reversed trend is observed in Pt@Activated carbon and Pt@carboHIPE, where the nanocomposites showed higher selectivity towards CO, but only slightly increasing selectivity towards CH₄ at higher temperatures. It should be noted that selectivity towards

Fig. 4 CO₂ consumption rate of various Pt@carbon nanocomposites at different temperatures (a), and time on stream study at 873 K (b)

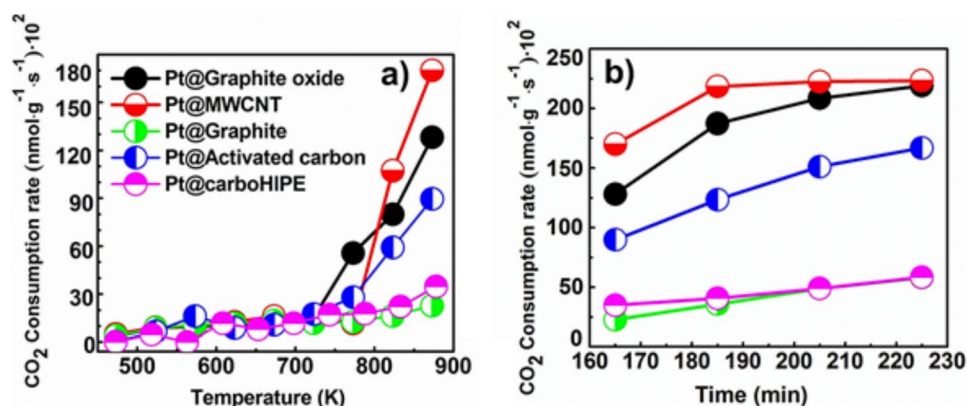


Fig. 5 Selectivity (♦ CO; ◇ CH₄) for the catalysts under review: **a** Pt@Graphite oxide, **b** Pt@MWCNT, **c** Pt@Graphite, **d** Pt@Activated carbon, and **e** Pt@carboHIPE

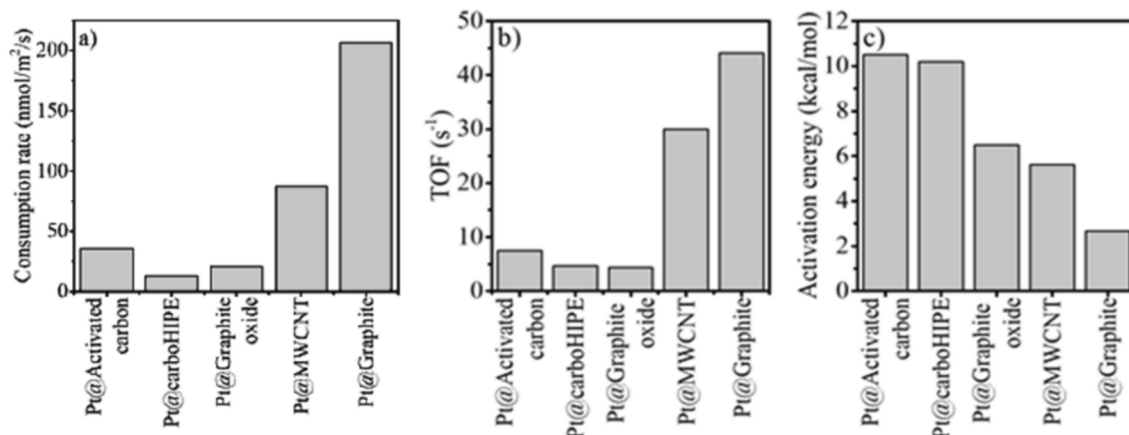
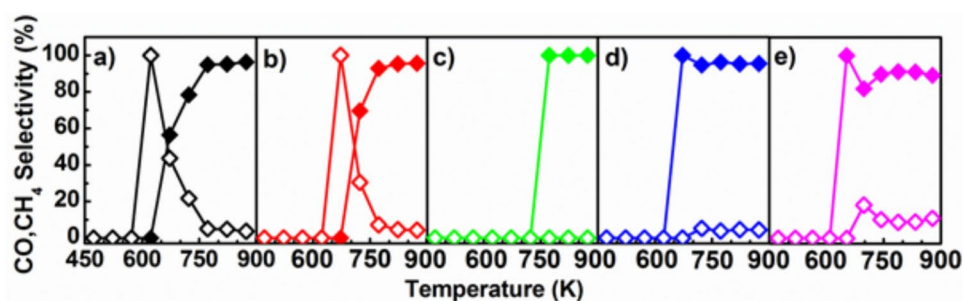


Fig. 6 CO₂ hydrogenation performance of the evaluated catalysts. **a** Consumption rate at 873 K, **b** TOF calculated at 873 K, and **c** activation energy

CH₄ remains low with increasing temperature for Pt@Activated carbon and Pt@carboHIPE, and reaches only 8% and 15% above 750 K, respectively (Fig. 5d, e).

It is known from the literature, that the number of defect sites in a carbon support can affect its catalytic performance, as larger number of defect sites allows for higher CO₂ adsorption capacity [37]. Similarly, the defects within the carbon matrix can also act as anchoring sites for Pt-NPs, allowing for its better dispersion on the surface of the support. This, in turn results in larger Pt-to-carbon interface, which then enhances the reaction with CO₂ to form CH₄ after dissociation of H₂ on the Pt surface [38]. In addition, it is possible to determine the degree of graphitization for each carbon support from Raman spectroscopy. The calculation is based on the ratio of the integral intensity of the D band and the G band, and it is suitable for any laser line in the visible range [39]. Moreover, the general equation developed by Cancado et al. (2006) can be used to describe the crystallite size (*La*):

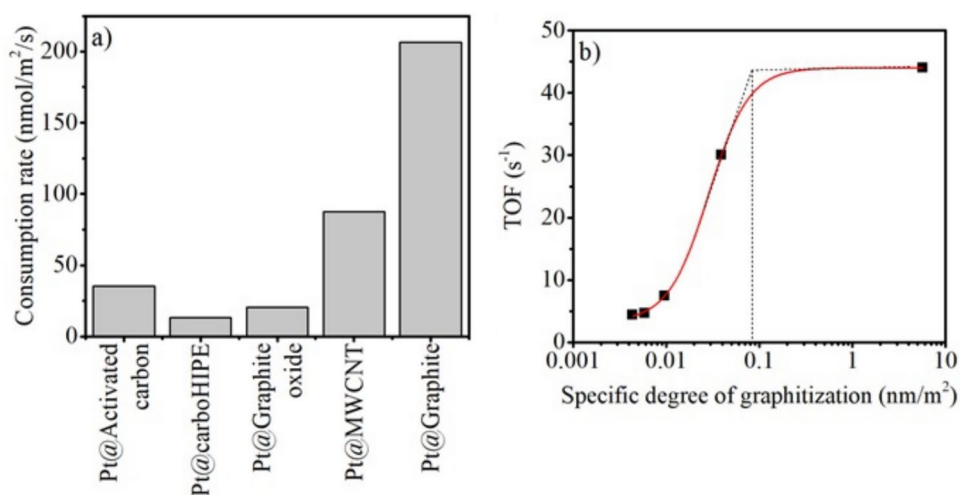
$$La \text{ (nm)} = 2.4 \cdot 10^{-10} \cdot \lambda^4 \cdot \frac{1}{I_D/I_G} \quad (4)$$

Good agreement was found with crystallographic studies (Table 1) and has recently been used to characterize the carbon-platinum interface [24]. Additional characterization

for the catalysts after reaction has been included in the supporting information.

To compare the catalytic performance of Pt@C nanocomposites, turnover frequency (TOF) values, defined as the number of moles consumed per active Pt site per unit time, at 673 K and 873 K were calculated and plotted in Fig. 6b. Here, the entire surface area is considered available and each atom on the surface is considered as an active site. Our experimental design with fixed loadings of Pt nanoparticles per m² result in a fixed number of active sites per nm² and it allows a clearer evaluation of the Pt@carbon interface. The number of Pt-NPs available at the surface of the nanocomposite is obtained from the size of the Pt nanoparticles and the amount of Pt added during the preparation of the catalysts. The number of active sites is calculated using the value 0.1521 nm²/2 atoms. Finally, the CO₂ consumption rate is divided by the number of active sites calculated and represents the number of molecules reacting per active site per second (Fig. 6a). One order of magnitude difference in TOF_{873 K} was found between graphite oxide and graphite. The activation energy was also calculated for each catalyst (Fig. 6c), which showed good agreement with the TOF results of 2.65, 5.62, 6.48, and 10.51 kcal/mol for Pt@Graphite, Pt@MWCNT, Pt@Graphite oxide, and Pt@Activated carbon, respectively. Similarly, the activation energy for graphite was 3.8 times lower than that for activated carbon.

Fig. 7 **a** Consumption rate at 873 K for the evaluated catalysts and **b** TOF vs specific graphitization degree



The large differences in TOF values can be explained by the degree of graphitization (Fig. 6b). The results show that the low degree of graphitization is related to lower activity (TOF_{873K} for Pt@Graphite oxide = 4.35), while higher graphitization led to higher turn-over frequencies of 30 and 44 s⁻¹ for MWCNT and graphite, respectively. The number of layers in graphitic samples have a strong correlation with their electronic properties [40]. Therefore, it is expected that the polarization of the Pt-NPs varies due to the charge transfer between the Pt clusters and the graphitic substrates. Density functional theory calculations by Ramos-Sanchez et al. (2013) describes the polarization process for Pt-NPs (~1 nm diameter) on graphite. Platinum atoms at the interface are positively charged, while the upper superficial layers carry negative charges, inducing clear differences in their DOS near the Fermi level along with differing catalytic properties [41]. Similarly, the degree of graphitization has been associated with stability and catalyst activity in electrochemical fuel cells [24]. In optics, Sasaki and Hitachi (2020) recently demonstrated the influence of the number of layers on light absorption. As a conclusion, N-layered graphitic materials with N > 1500 can be considered as graphite in light absorption from the infrared to the visible range [42].

As described in the experimental section, the nanocomposite catalysts contain a fixed loading of 25 μg m⁻² Pt NPs per unit surface area of the carbon support. Thus, the number of Pt-NPs varies between samples depending on the surface area of the carbon support. Figure 7a shows consumption rates that are proportional to the amount of Pt-NPs, which better explains the trends observed in Fig. 6.

Finally, we propose the so-called ‘Specific Degree of Graphitization (SDG)’ as a characteristic of the Pt-C interface in heterogeneous catalytic reactions. The SGD index is defined as the ratio between the crystallite size obtained by the Scherrer equation from the graphitic peak (corresponding to the (002) reflections), and the surface area

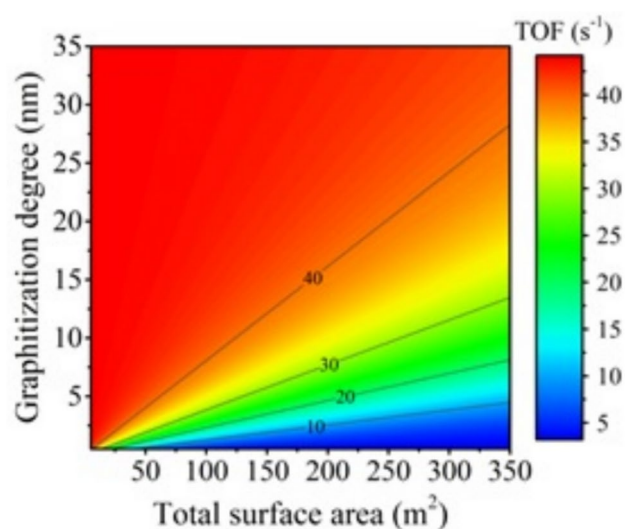


Fig. 8 Dependence of CO₂ hydrogenation activity of Pt@carbon catalysts on the total specific surface area and the Specific Degree of Graphitization (SDG) of the carbon supports

of the carbon support (crystallite size/surface area in [nm m⁻²] or [nm⁻¹]). The input data for the calculation and the resulted SDGs are listed in Table 1 and depicted in Fig. 7b in a semi-logarithmic graph by plotting TOF values against the logarithm of the corresponding SDGs of each catalyst on different carbon supports.

The resulted sigmoid-type curve in Fig. 7b was fitted by a logistic function to guide the eye, where the inflection point for high TOF performances of Pt NPs on carbon substrates can be found at SDG ≈ 0.085 nm m⁻². To further visualize the carbon support-dependent catalytic performance of the evaluated Pt-NPs@carbon nanocomposites in gas-phase CO₂ hydrogenation, a contour plot was constructed in Fig. 8, where the variation of TOF on the total specific surface area and the graphitization degree can be seen. This representation confirms, that although the specific surface area is an important parameter, detailed description of the

CO₂ hydrogenation activity on carbon supported catalysts cannot be realized without taken the degree of graphitization of the support into consideration. A great example to that is, that although the graphite-based catalyst had the lowest S_{BET} (Table 1), it showed the highest TOF value in our study as shown in Fig. 7b and 8. Based on the above findings, we propose the use of high surface area carbon supports with an optimal graphite stacking thickness of 30 layers (10 nm thick layer built up from C (002) layers with 0.34 nm plane spacing) for gas-phase CO₂ hydrogenation on Pt@carbon catalysts.

The mechanism of the reverse-gas shift reaction (RWGS) (Eq. 1) is still under scrutiny in the literature, and after the initial CO₂ adsorption step the following three different elementary steps were proposed:



CO₂ hydrogenation seems to follow one of the above pathways depending on the catalyst, the support, and the reaction conditions. It is generally assumed that RWGS proceeds through the carbonate-mediated mechanism on Pt, as CO₂ dissociation is only favorable on metals of high oxygen affinity (Rh, Ni, Cu) and the high stability of the surface formate makes it a spectator group [43]. However, there are contradictory reports on the possible adsorption and dissociation of CO₂ on Pt. Early UHV study demonstrated that CO₂ adsorbs and dissociates on Pt(111) only upon K doping [44], whereas UHV XPS measurements showed the presence of chemisorbed CO on Pt foil [45]. Recent APXPS results found evidence on the adsorption and dissociation of CO₂ to CO on Pt(111) single crystal in the presence and absence of additional H₂ even at room temperature [36].

Catalysts' functionalities can be described by the energetics of the adsorbed species, nevertheless extending the study from well-defined single crystal model systems to supported nanoparticle catalysts is challenging introducing [46]. DFT calculations showed that CO₂ hydrogenation goes entirely through the carbonate pathway, and the low RWGS activity on self-standing Pt NPs is associated with the weak adsorption of CO₂ in the initial step. The latter favors CO₂ desorption before hydrogenation takes place, and thus by stabilizing adsorbed CO₂ on the catalyst surface increases conversion. It is known that catalyst supports can fundamentally influence a catalyst's behavior by, e.g., perturbing the electronic properties of the metal nanoparticle, stabilizing certain surface species etc. For instance, low-coordinated Pt sites at corners and edges adsorb CO₂ primarily,

but the weak bonding between these sites and the adsorbate hinders to achieve high activity. Whereas, e.g., surface –OH groups in Pt/SiO₂ or oxygen vacancies in Pt/TiO₂ can enhance CO₂ adsorption on Pt NPs [47]. By choosing the right support the electronic structure of the interface with the catalyst particle can be tailored, too. Electronic metal-support interaction (EMSI) describes chemical bonding and the associated charge transfer at the metal/support interface. This, in turn, charges the metal either negatively or positively, where the resulting electron-rich or electron-deficient surface becomes donor or acceptor for the reactants. Carbon materials are tunable supports, where modulating defect density, surface chemistry (e.g., the density of Electron-Withdrawing Groups (EWG) like ether, carbonyl, and carboxyl, and Electron-Donor Groups (EDG), like hydroxyl), and heteroatom dopants can tailor electron transfer via moving electrons through the π -bonds of the carbon structure (resonance) or bond polarization by electronegativity differences (inductive effect). It is known that adsorption is highly influenced by the energy and filling of metal d-band, and electron transfer from the metal particle to the support shifts the d-band of the metal upward closer to the Fermi level (shallow d band). This results in stronger intermediate-surface interaction [48]. Such strongly surface-bound CO is the origin of CO-poisoning of Pt, which hinders, e.g., the CO oxidation reaction. Several studies investigated the promotion of the latter on Pt using various carbon supports with fine-tuned electronic properties, changing the surface density of defects and oxygen-containing functional groups (OCGs). It was concluded, that increasing EWG/EDG ratio and higher defect density empty and lower the Pt d-band, which weaken the CO–Pt adsorption on the carbon black-carbon nanotube-carbon nanofiber series [[49, 50]], and on carbon nanotubes with tailored EWG/EDG ratio [51]. In our case, for stronger CO₂ adsorption shallower d-band, i.e. filling of the Pt d-bands via electron transfer from the carbon support to the Pt and the corresponding upward shift of the d-band center, is required. It is also known that d-band centers are shallower for higher index Pt surfaces, whose CO₂ adsorption energy is higher than those on planes of lower Miller indices [52]. Moreover, CO₂ weakly physisorbs on uncharged Pt, whereas CO₂ molecule chemisorbs on charged Pt surfaces through partial electron transfer from Pt to the antibonding $2\pi_u$ LUMO to form CO₂[−]. This reduces bond strength and bends the CO₂ molecule, and size-dependent CO₂ adsorption on Pt clusters found evidence of both molecular and dissociative adsorptions. It was suggested for practical applications to deposit small Pt clusters onto electron donating supports, like metal oxides or carbon [53]. To that end, a decreasing EWG/EDG ratio needs to be realized by either increasing EDG or decreasing EWG surface density. Here, higher SGD (Table 1) implies

the increase in sp^2/sp^3 , which in turn means the decrease of the electron withdrawing potential of the support. Furthermore, Raman results in Figure S2-6 indicated the formation of defect in the carbon supports, which can further stabilize the adsorbed CO₂ molecules at the Pt/C interface, as discussed above [54].

4 Conclusions

Pt supported on graphite oxide, carboHIPE, activated carbon, MWCNT, and graphite were synthesized to investigate the influence of the physicochemical properties of the carbon support on the performance as a nanocomposite catalyst in CO₂ hydrogenation reaction in gas phase. No significant effect of the carbon support on Pt nanoparticle size distribution was seen, as NP size was mainly controlled by the synthesis method. The nanocomposite catalysts were prepared with identical Pt-NPs loading of 25 $\mu\text{g m}^{-2}$ of support and their thermal stability was characterized by TGA and Raman spectroscopy at reaction temperatures. The carbon supports were found to have different amounts of defective sites in the carbon skeleton, as shown by the Raman results, which in turn affect the platinum–carbon interface and the electronic interactions that determine catalytic activity.

A relationship was found between the degree of graphitization of the carbon support and the catalytic performance of the studied Pt@carbon catalysts. We, therefore, defined the so-called Specific Degree of Graphitization (SDG), with which we were able to describe the correlation between the CO₂ hydrogenation activity (TOF at 873 K) and the degree of graphitization of the carbon support in Pt@carbon composites. The highest activity among the investigated catalysts was found for the Pt on graphite nanocomposite, showing a TOF_{873 K} as high as 44 s⁻¹ along with the highest SDG index of 5.7130 nm m⁻². The higher CO₂ hydrogenation activity in RWGS originates from the increasing electron donation of the carbon support to the Pt nanoparticles by the increase in graphitization. This, in turn, enhances the rate-limiting initial CO₂ adsorption on Pt by shifting its d-band center towards the Fermi level, demonstrating enhance CO₂ activation. We propose the defined SDG index as a powerful indicator for the prediction of CO₂ hydrogenation catalytic activity on Pt@carbon nanocomposites.

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